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# Controlling the Organization of PEDOT:PSS on Cellulose Structures

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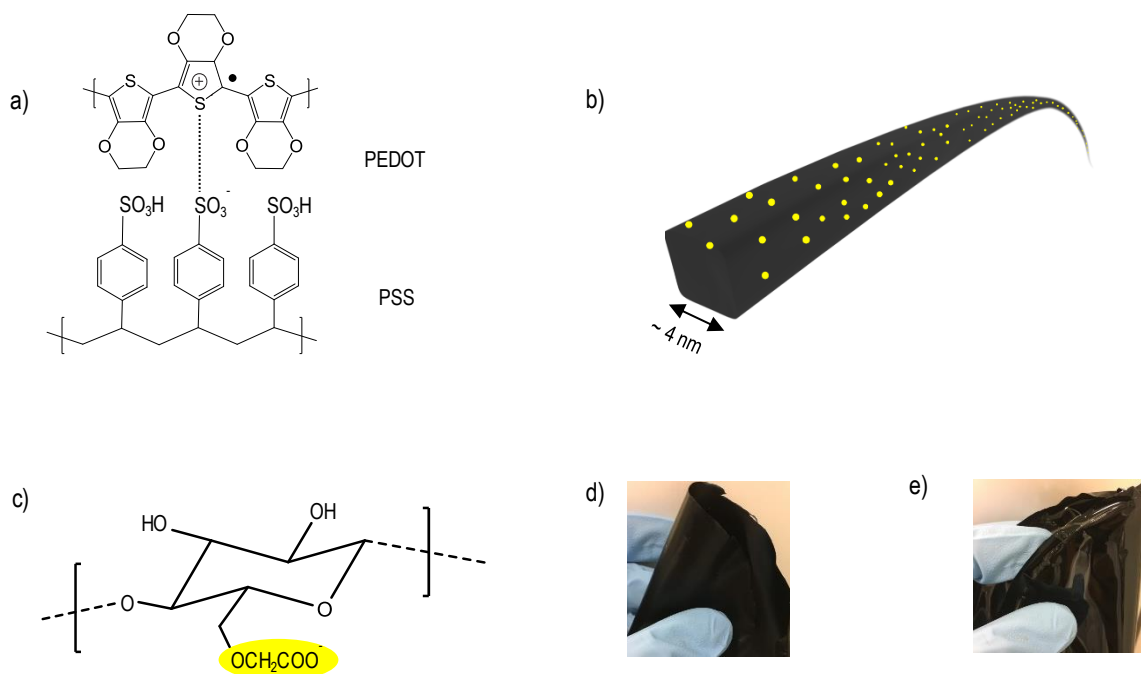
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Composites of biopolymers and conducting polymers are emerging as promising candidates for a green technological future and are actively being explored in various applications, such as in energy storage, bioelectronics, and thermoelectrics. While the device characteristics of these composites have been actively investigated, there is limited knowledge concerning the fundamental intra-component interactions and the modes of molecular structuring. Here, using cellulose and poly(3,4-ethylenedioxythiophene):poly(styrenesulfonate) (PEDOT: PSS) it is shown that the chemical and structural make-up of the surfaces of the composite components are critical factors that determine the materials organization at relevant dimensions. AFM, TEM and GIWAXS measurements show that when mixed with cellulose nano-fibrils, PEDOT:PSS organizes into continuous nano-sized bead-like structures with an average diameter of 13 nm on the nano-fibrils. In contrast, when PEDOT:PSS is blended with molecular cellulose, a phase-segregated conducting network morphology is reached, with a distinctly relatively lower electric conductivity. These results provide insight into the mechanisms of PEDOT:PSS crystallization and may have significant implications for the design of conducting-biopolymer composites for a vast array of applications.

At this time of rapid human development, environmentally friendly means of addressing the world's increasing technological demands are needed more than ever. Composite systems of electronically active polymers and renewable biomaterials are seen as one of the key solutions in this drive towards a green and clean planet.<sup>1-5</sup> Conducting polymers are frequently blended with biopolymers for an array of novel applications where the biopolymers provide hierarchical

flexibility,<sup>6</sup> improve performance,<sup>7</sup> mechanical stability,<sup>8</sup> processability,<sup>9</sup> and biocompatibility<sup>10</sup> to the pure conducting polymer, leading to powerful synergistic performance and device functions. For example, the production of stand-alone bulk supercapacitor electrodes has recently been enabled via an amalgamated system including the conducting polymer poly(3,4-ethylenedioxythiophene):poly(styrene-sulfonate) (PEDOT:PSS, Figure 1a) and the mechanically strong cellulose nanofibril (CNF, Figure 1c).<sup>8</sup> The resulting conducting paper, called *power paper*, is flexible, scalable and exhibits excellent mechanical and electrical properties including high mixed ionic and electronic conductivity. The addition of cellulose to the PEDOT:PSS enables us to manufacture sustainable, thick and bulky supercapacitor electrodes, which is not possible to achieve using PEDOT:PSS only. However, notwithstanding the promising properties of such systems, there is a lack of clear structural paradigms guiding the selection of compatible conducting polymers and biopolymers.



**Figure 1.** a) Chemical structure of PEDOT:PSS b) schematic of CNF with carboxymethyl groups (yellow) c) chemical structure of carboxymethylated cellulose (CMC) with degree of substitution (DS) 1. d) *power paper* (CNF-PEDOT:PSS conducting paper) e) CMC-PEDOT:PSS conducting paper.

Wood is the most readily available and environmentally friendly biomaterial.<sup>1</sup> Wood has a hierarchical structure which makes it an essential source of an array of biomaterials of different sizes and chemical compositions. One of the main components of wood, cellulose, has a long list of commendable attributes and is commercially available in a vast array of different structural forms.<sup>11, 12</sup> The anisotropic micron to millimeter sized fibers that constitute a regular sheet of paper can be mechanically and/or chemically treated to give highly anisotropic cellulose nanofibril (CNF).<sup>13</sup> CNF (Fig. 1c) is being investigated in several applications within bioactivity,<sup>14</sup> gel formation,<sup>15</sup> transparent substrates,<sup>16</sup> and composite structures.<sup>17</sup> CNF is often prepared with some of its hydroxyl groups substituted with functional ionic groups such as carboxymethyl groups to facilitate dispersibility in water.<sup>18</sup> The degree of substitution (DS-value) quantifies the number of hydroxyl groups per monomer unit that have been substituted with a charged group. Typically, CNF will have a DS between 0.05 and 0.35. It should be noted that, as the substituted charges reside only on the surface of the CNF, the surface charge density of the fibril is high. For a DS of 0.35 the average distance between adjacent charged groups is  $\sim 0.7$  nm. Beyond a DS of 0.4 the glucan chain becomes so highly charged that it will dissolve into the polyelectrolyte form, i.e. carboxymethyl cellulose (CMC, Figure 1b).<sup>19</sup> Owing to its physiological inertness and high water solubility, polymeric CMC has been in use for a long time as a viscosity regulating food additive and, recently, CMC has also been actively investigated as a safe binder material in state-of-the-art batteries and supercapacitors.<sup>20</sup>

The stability and high mixed electronic-ionic conductivity of PEDOT:PSS (Fig. 1a) has made it the commercially most successful conducting polymer,<sup>21, 22</sup> even though several high-performing alternatives can be found in the various families of conjugated polymers.<sup>23, 24</sup> The polyanion PSS neutralizes and dopes the charged PEDOT, while excess PSSH in the system makes the whole material dispersible in water and provides easy processing.<sup>25</sup> PEDOT:PSS's mixed ionic/electronic charge transport capability has opened up an array of applications, such as organic electrochemical transistors and organic electronic ion pumps.<sup>26</sup> Recently, PEDOT:PSS is being intensively investigated for energy generation and storage applications, such as in fuel cells, thermoelectrics and supercapacitors.<sup>8, 27-30</sup>

The morphology of PEDOT:PSS has actively been studied with a range of techniques.<sup>31-33</sup> In solution state, PEDOT:PSS is found as a nano-suspension with ~ 30 nm size particles and 500 - 600 nm size aggregates of the particles. Individual particles are composed of an inner core of PEDOT-rich region surrounded by a soft PSS-rich region.<sup>34-36</sup> On the other hand, dried and thin-films of PEDOT:PSS are depicted as segregated PEDOT-rich and PSS-rich regions. The PEDOT-rich regions are composed of crystalline grains of a few nanometers size, organized into clusters of tens to a hundred nanometers depending on the details of the sample preparation.<sup>31, 37, 38</sup> It has been reported that so-called secondary dopants such as dimethyl-sulfoxide (DMSO), ethylene glycol, and methanol have an impact on the phase separation and organization of PEDOT and PSS. This phase separation results in a highly interconnected PEDOT network with stronger interchain coupling of the PEDOT (PEDOT  $\pi$ -stacking), thereby improving the electronic conductivity significantly.<sup>39-42</sup> The effect of the substrate on PEDOT:PSS morphology is largely unexplored but recent theoretical studies suggest that the interaction of PEDOT:PSS with the substrate is critical for its organization and to promote swift charge transport.<sup>43</sup>

PEDOT:PSS-based self-standing conducting paper can be made from either CMC or CNF for energy storage applications (Figure 1d and e). Both kinds of conducting papers are mechanically resilient to continuous folding and rolling. However, the fibril-based papers show improved electrical device characteristics as compared to their polymeric counterparts, indicating either a different structuring, or differences in the interaction between PEDOT:PSS and cellulose, or both. It has also been shown that the level of adsorption of PEDOT:PSS on cellulose is dependent on the pH of the system.<sup>44</sup> PEDOT:PSS shows maximum adsorption on cellulose at its natural pH levels, which are in the range of 2 due to the low pKa of the sulfonic acid groups. At such low pH levels, the charged groups of the cellulose are protonated, and the charged fibrils become neutral. Hence, the electrostatic repulsion between the cellulose and the negatively charged sulfonate groups of the PEDOT:PSS is at its minimum and Van der Waals and dipolar interactions dominate. This means that at low pH levels the interactions of cellulose with PEDOT:PSS are not primarily dictated by the DS of the cellulose. The superior device properties observed in the fibrillated samples seems to suggest that the geometrical structure of the cellulose plays a critical role in PEDOT:PSS organization.

Even though earlier investigations have to some degree addressed the interactions between the constituents of a composite and the structural organization, there is only limited knowledge of the hierarchical arrangement of composites from the molecular to the micrometer size scale.<sup>45-48</sup> Therefore, despite the generic interest in blend systems composed of conducting polymers and biomaterials, there is a lack of clear understanding of the structural organization of these composites and its impact on device performance. This is partly due to the complexity of the composite, making sample preparation, measurement and data analysis a challenging task.

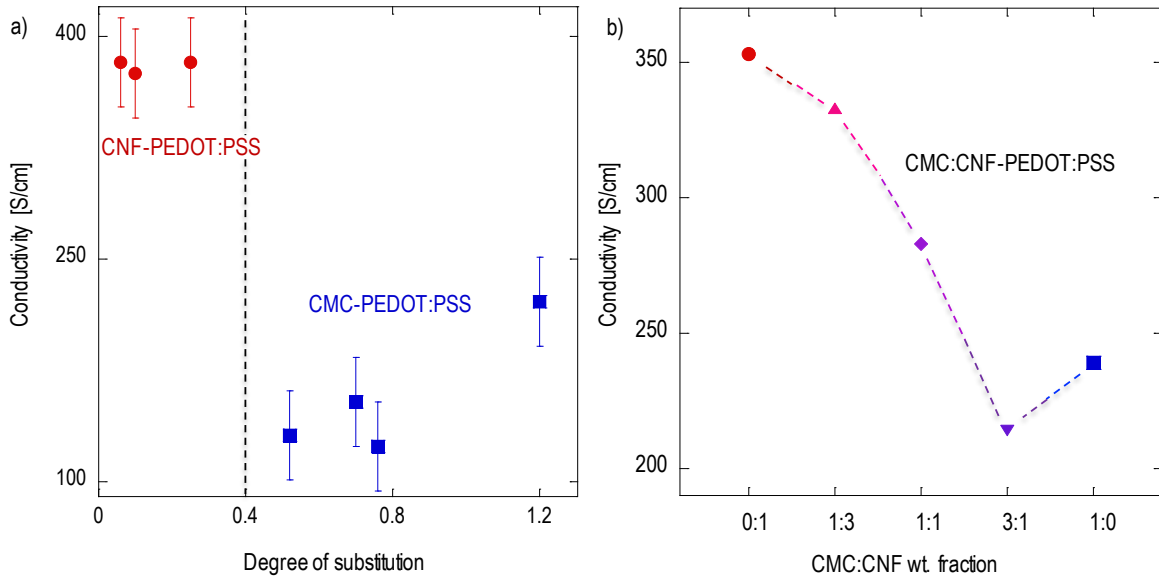


Here, by using two structurally different cellulose systems in a composite system with PEDOT:PSS, we show that the PEDOT:PSS organizes itself in nanometer-sized bead-like structures on top of CNF nanofibers with predominantly pi-stacked PEDOT, whereas CMC brings about an organization which is dominated by phase separation and stronger lamellar stacking of PEDOT. The structural variation is also consistent with a distinct improvement in electronic conductivity in the fibril-based systems. We use Wide Angle X-ray Scattering (WAXS) for crystallographic and textural investigations and direct measurements with Atomic Force Microscopy (AFM) and Transmission Electron Microscopy (TEM), to show the distinct morphology and crystallographic organization of the conducting polymer in the different cellulosic systems. The necessity of understanding and tailoring the structure and chemistry of the constituents of bio-composites for more efficient devices is emphasized.

## **RESULTS AND DISCUSSION**

The conductivity of the power-papers was studied with four-probe electrical measurements. Figure 2a shows the variation of conductivity with the degree of carboxymethylation (DS) of the cellulose. Fibrils (CNF) correspond to DS values below 0.4 while above this value the cellulose is found in the form of a polymeric CMC. We observe two distinct regimes with respect to the amount of carboxymethyl charge on the cellulose. In the low-charge, fibrillated zone, the conductivity is higher with values in the range of 400 S/cm. At higher charge contents, where the cellulose is polymeric and completely soluble in water, the conductivity is lower (150 S/cm). The fact that the conductivity values stay relatively constant within these two regimes suggests that the PEDOT organization is not varying markedly in the respective regimes but that there is distinct conductivity level difference between the two material systems.

By studying previous literature, which reports the effect of adding excess of PSS to PEDOT:PSS, we can compare the role of the non-conducting fibrils in the conductivity of the paper with the role of a non-conducting polymer (PSS).<sup>49, 50</sup> The amount of the electronically conducting component, PEDOT, in the *power paper* is only around 20% by weight, and the remaining component is the non-conducting cellulose together with PSS. A renormalized conductivity with respect to the weight of PEDOT gives a value of about 2000 S/cm for the PEDOT-component in the fibril-based paper. This value is more than half of the renormalized conductivity of PEDOT:PSS which is 3500 S/cm at 1:2.5 PEDOT:PSS ratio.<sup>51</sup> However, if we replace the amount of cellulose in power paper by PSS, the renormalized conductivity is then less than 700 S/cm. It is then clear that the addition of excess PSS has a more detrimental effect on the conductivity than the addition of fibrils to PEDOT:PSS. It can also be seen that the replacement of CNF by CMC in power paper also has a similar effect as adding excess PSS.



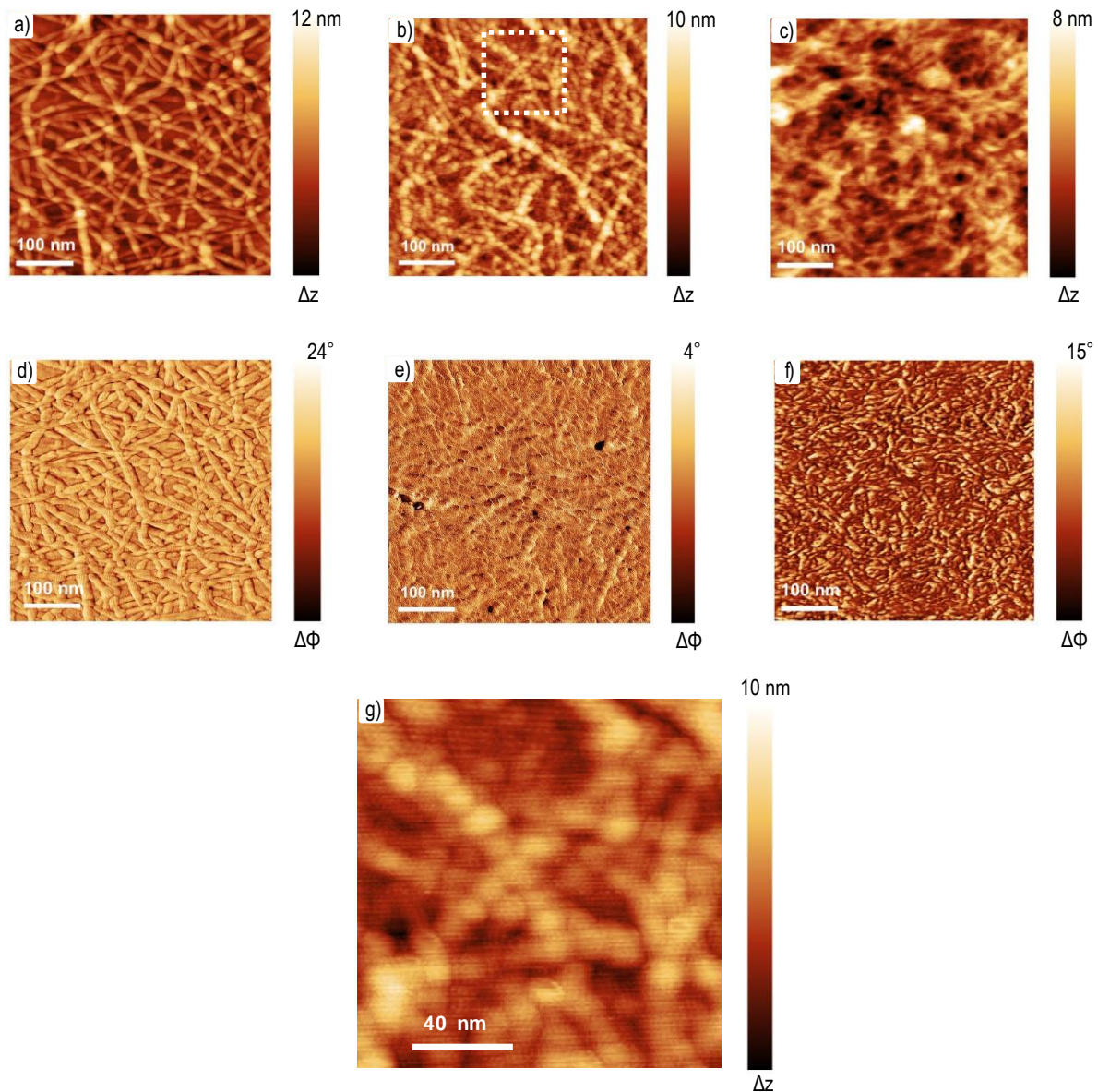
**Figure 2.** a) Conductivity of CNF-PEDOT:PSS and CMC-PEDOT:PSS conducting papers b) Conductivity of conducting paper at different CMC:CNF weight fractions.

We further investigated the role of the cellulose in determining the conductivity of the PEDOT by mixing CNF and CMC with PEDOT:PSS to prepare the conducting paper. Figure 2b displays the electrical results from experiments where the addition of CMC to the CNF brings about a decrease in the conductivity of the conducting paper. The conductivity decreases until the ratio of CNF:CMC is 1:3 and then increases slightly for the pure CMC-PEDOT:PSS sample, but still at a level much lower than for pure CNF-PEDOT:PSS. Hence, an addition of a small amount of CNF to a CMC solution, or vice versa, leads to a significant drop of conductivity. This result suggests that the pure CNF and CMC blends promote totally different kinds of PEDOT organization. An addition of a small amount of CNF to CMC-PEDOT:PSS results in disorder in the charge transport pathway and leads to a decrease of conductivity. However, with more CNF in the system the CNF-governed organization of PEDOT takes precedence. Since this organization leads to higher conductivity, the conductivity of the blend system increases until it reaches the maximum conductivity for blends made with pure CNF-PEDOT:PSS.

Charge transport in PEDOT:PSS films is mainly governed by the ease of charge transfer between clusters of PEDOT:PSS.<sup>31, 52, 53</sup> The PEDOT:PSS film is understood to be a phase-separated mix of PEDOT-rich clusters surrounded by PSS-rich lamellae.<sup>54</sup> Nardes et al report that a thermally activated charge transport dominates at room temperature and that the anisotropy introduced by phase separation leads to large variations in the conductivity values in the lateral and perpendicular directions.<sup>31</sup> The differences in conductivity between the CNF-based and the CMC-based samples indicate that there is a marked difference in the charge transport mechanism and, consequently, in the organization of the PEDOT of the two blend systems. In what follows, AFM and WAXS measurements show that there is a more pronounced  $\pi$ -stacking and improved long range domain organization in the CNF samples, which are both known to lead to higher conductivity.<sup>37, 55</sup>

Even though AFM is a surface sensitive technique which typically cannot give details of bulk morphology, it is a powerful method to clarify the influence of interactions between the components of a composite system, and, in combination with other techniques, can provide critical information of the phase separation and organization of the material. The topographic AFM image of the pure CNF in Figure 3a shows well dispersed, high aspect-ratio fibrils. Since the fibrils are smaller than the lateral resolution limit, of around 5 - 10 nm dictated by the tapping-mode AFM tip, the vertical profile was used to measure the dimension of the fibrils. It was determined that the fibrils have diameters in the range of 3-5 nm. When the CNF-PEDOT:PSS composite is examined (Figure 3b), we see evidence of bead-like PEDOT:PSS structures aligned along the fibrils. The zoom-in image of the center part (Figure 3g) shows in detail how spherical clusters of PEDOT:PSS form a continuous bead-path. To extract the size of the beads, the length of a series of beads on a single fibril was averaged. On an average, the beads are  $13 \pm 1$  nm in diameter. However, we observe a few outliers with diameters as large as 25 nm. These nano-beads have a size which is in agreement with dynamic light scattering (DLS) measurements of PEDOT:PSS particles in the solution state.<sup>34, 35</sup>

To investigate the nature of the beads, the phase image of the scans was studied. From previous phase image studies of PEDOT:PSS it is known that PSS, which adsorbs water and is a relatively softer material than PEDOT, has a lower phase angle and appears thus as relatively darker.<sup>56</sup> Regions which are PEDOT-rich are represented by brighter areas in the image. In Figure 3e it is seen that the individual beads have varying darkness suggesting that the PSS coverage is non-uniform. Based on models of PEDOT:PSS, which predict a PSS-rich outer cladding and a PEDOT-rich core, we suggest that the observed beads are PEDOT:PSS particles with an irregular PSS-rich outer region and an inner PEDOT-rich region.

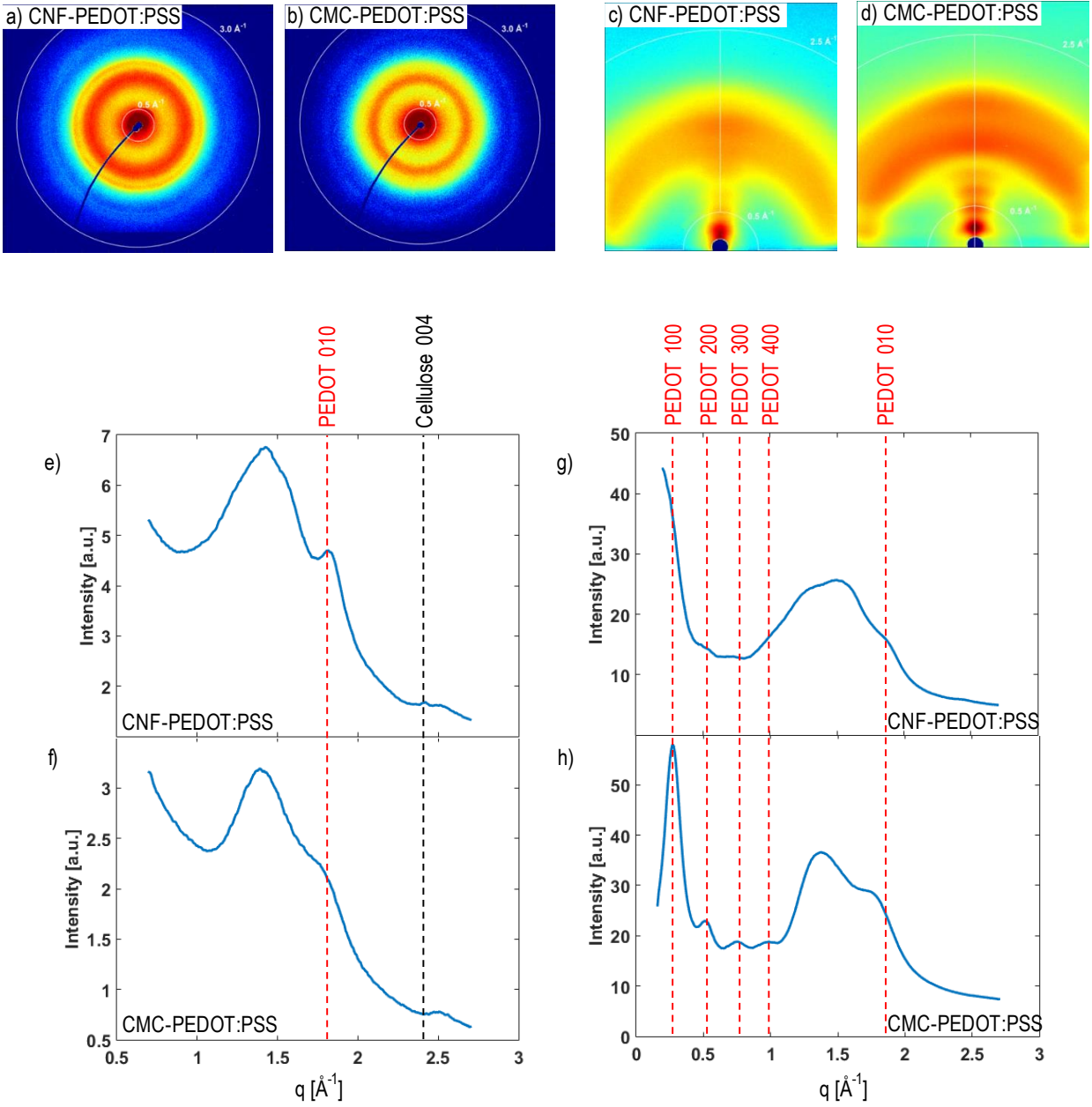


**Figure 3.** AFM topography and phase images of a) and d) pure CNF, b) and e) PEDOT:PSS on CNF, c) and f) CMC-PEDOT:PSS, respectively. g) a zoom-in image of the area marked with a dotted rectangle in b) (PEDOT:PSS on CNF).

Figure 3c displays the topography of CMC-PEDOT:PSS. The surface is a mix of a few globular structures with sizes in the range of 40-60 nm and some elongated structures with lengths 75 – 95 nm. The phase image (Figure 3f) that was simultaneously taken shows a phase separated brighter

phase which is a mix of globular and elongated structures. As in the previous phase image, the bright parts depict the PEDOT-rich regions. The CMC, similar to PSS, is hygroscopic and is expected to swell at ambient conditions, hence becoming softer than its surrounding and thus have a relatively lower phase (dark region). In support of this interpretation, the phase difference observed in a CMC-PSS mix is minimal (Supporting Information, Figure S1d) indicating that they indeed have similar mechanical properties at ambient conditions. The phase image in Figure 3f shows a reduction in the interconnectivity of the PEDOT-rich regions as compared to a pure PEDOT:PSS film (Supporting Information, Figure S1c). Therefore, it appears that CMC brings about a disruption in the phase segregation process. Even though the PSS and CMC segregate, giving rise to PEDOT-rich regions, the PEDOT-rich regions are not organized into interconnected structures to the extent as found in pure PEDOT:PSS. This reduction in the interconnectivity of the PEDOT-rich regions is likely the cause for the higher drop of the conductivity in the CMC-PEDOT:PSS blends.

To establish the form of PEDOT organization, WAXS measurements in transmission (with the X-ray beam impinging perpendicular to the film surface) and in Grazing Incidence (GIWAXS, with the X-ray beam nearly parallel to the film surface) were performed. WAXS is a suitable technique for probing the structural ordering of polymers from a few to several hundred nanometers.<sup>57</sup> The transmission and GIWAXS modes can be used complementarily to derive a semi 3-dimensional representation of the investigated samples.



**Figure 4.** WAXS in transmission incidence for conducting paper made with PEDOT:PSS and a) CNF b) CMC and GIWAXS in parallel incidence c) CNF and d) CMC. The corresponding azimuthal integration for transmission incidence for e) CNF f) CMC and parallel incidence g) CNF and h) CMC.

Figure 4 shows WAXS data of free-standing films made from PEDOT:PSS mixed with CNF or CMC. It was noted that the WAXS data for CNF samples of different DS were qualitatively similar to one another. The same observation was made for CMC samples of different DS. Therefore, patterns collected in the transmission and parallel configuration for the samples DS 0.12 and DS 1.2 are shown as representative examples of fibrillated and soluble samples, respectively. WAXS data for the other charge groups can be found in the supplementary information. For ease of comparison, the WAXS pattern data of all samples has been azimuthally integrated and the intensity plotted with respect to the length of the scattering vector,  $|q| = 4\pi \sin\theta/\lambda$ , where  $2\theta$  is the scattering angle and  $\lambda$  is the X-ray wavelength.

From the transmission measurements of the fibrillated samples (Figure 4a and 4e) we observe that PEDOT on CNF has  $\pi$ - $\pi$  ordering as can be seen from the PEDOT  $010$  peak at  $q = 1.80 \text{ \AA}^{-1}$ , corresponding to a real space distance of  $3.5 \text{ \AA}$ , as has been reported previously.<sup>23, 51</sup> The collected diffraction patterns (Figure 4a) shows continuous rings indicating PEDOT  $010$  stacks randomly oriented with respect to the film surface normal (fiber symmetry), i.e. edge-on orientation (Supporting Figure S4 illustrates the different types of PEDOT packing). The cellulose  $012$  and  $020$  rings observed at scattering vector  $q = 1.41 \text{ \AA}^{-1}$  and  $q = 1.51 \text{ \AA}^{-1}$ , respectively, demonstrate that the fibrils are randomly oriented too. The grazing incidence pattern (Figure 4c and 4g), on the other hand, gives information of the crystal packing along the substrate normal, and shows the additional population of a predominantly  $\pi$ - $\pi$  stacked PEDOT in face-on orientation as indicated by the bright  $010$  arc at  $90^\circ$  azimuthal angle (on the surface normal), with weak  $\pi$ - $\pi$  ordering at other angles. No significant lamellar order as can be seen in the weak scattering at small angles corresponding to the expected position of  $h00$  PEDOT peaks. Observing the AFM (Figure 3), it is reasonable to assume that the CNF takes the primary role in the organization of the PEDOT due



to the large surface area it provides. In summary, the free-standing *power paper* is composed of two populations of PEDOT covering the CNF with  $\pi$ - $\pi$  stacking occurring both face-on and edge-on with respect to the substrate. Such an organization is consistent with both AFM and WAXS data.

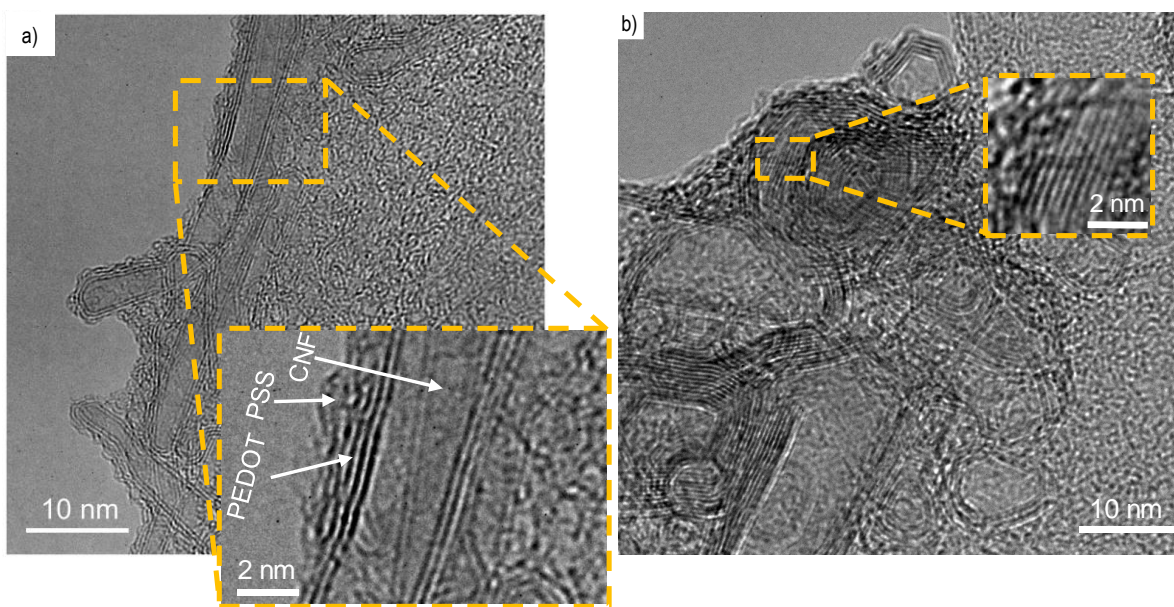
Molecular dynamic simulations (Supporting Figure S5) by Mehandzhiyski et al<sup>58</sup> of different simplified systems containing only PEDOT:PSS, one fibril and water, show that in certain systems PEDOT arranges itself in bead-like aggregates with a lower density region in-between the beads. The similarity to the observed bead-like structures in Figure 3 is noteworthy. In such systems the orientation of the PEDOT is predominantly face-on on top of cellulose fibrils. Furthermore, simulated diffraction patterns of the same system computed by the same authors show distinct PEDOT  $\pi$ - $\pi$  stacking peaks. Significantly, the simulated diffraction patterns did not indicate any lamellar structures of PEDOT and, taken as a whole, the simulations are thus strongly corroborating the experimental AFM and WAXS data reported here.

The transmission WAXS data of the CMC-PEDOT:PSS samples (Fig. 4b and 4f) show a very weak PEDOT  $010$  signal, which seems to indicate the absence of  $\pi$ - $\pi$  edge-on ordering of PEDOT. Contrary to these observations, in the GIWAXS measurements of CMC-PEDOT:PSS paper we detect very strong lamellar peaks which indicate the presence of edge-on PEDOT crystals. In addition, similar to the fibril containing samples, face-on stacked PEDOT is indicated by the arc at  $90^\circ$  azimuthal angle. Hence, the CMC-PEDOT:PSS samples have a mix of edge-on and face-on oriented PEDOT. We note that there are multiple factors influencing the organization of the PEDOT. Since this is a truly bulk material (sample thickness is several tens of microns) it is possible that the formation of PEDOT crystallites in the bulk is markedly different to the PEDOT crystallite formation at the surface in contact with the petri-dish during drying. Based on

computational studies, Franco-Gonzalez et al. report that amorphous substrates, such as the glass petri-dish substrate used in this experiment, promote a predominantly edge-on morphology in wet state and a mixed edge-on/face-on morphology in dry state.<sup>43</sup> Even though these theoretical models were done for PEDOT-Tosylate and do not contain cellulose, PSS, DMSO and glycerol, which are present in our films, these studies are noteworthy because they address the importance of the supporting substrate and the water content. Due to the wetness present due to the glycerol used as a plasticizer and to the hygroscopic nature of PSS and cellulose we can conjecture a mixed organization with a possible preference for edge-on orientation. It is possible that both edge-on and face-on orientations are present both at the surface and bulk of the free-standing paper. However, it is not clear which specific phenomenon dictates the PEDOT structuring in the bulk of the CMC-PEDOT:PSS samples.

To investigate the crystalline organization of PEDOT in the mixed systems we performed high-resolution transmission electron microscopy (HRTEM). Figure 5 shows overview TEM images of CNF-PEDOT:PSS solution cast on a TEM copper grid where a few fibrils on the edge of the film can be observed. Enhanced magnification image of one of the fibrils (Fig 5a inset) shows that surrounding it, there is a stack of crystals with a stacking distance  $\sim 4 \text{ \AA}$ . This coincides well with the  $\sim 3.5 \text{ \AA}$   $\pi$ -stacking distance of PEDOT crystals and suggests the observation of PEDOT crystals stacked face-on on the surface of the fibril. In addition, there is a thin, amorphous, non-uniform covering layer on top of the crystal stacks which we expect to be a PSS outer layer that is bound to the PEDOT. Furthermore, we observe that these crystalline stacks are not perfectly continuous on top of the fibril indicating that the PEDOT crystals may be organized on the surface of the fibrils in the nano-bead structures observed in the AFM images.

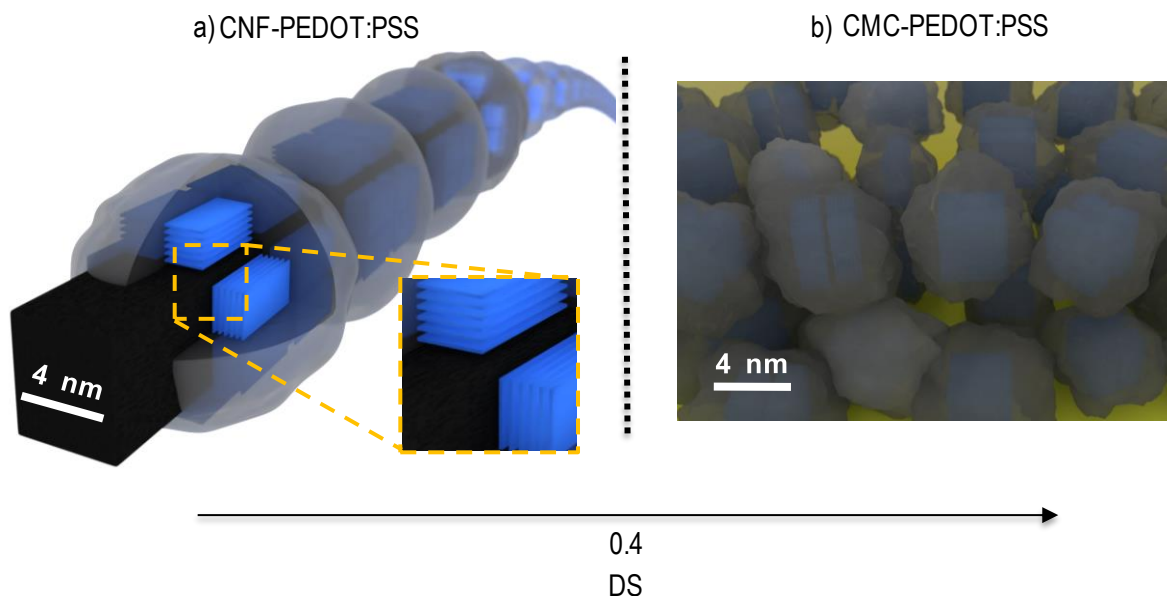
Figure 5b shows highly organized crystalline lattices with inter-lattice distance of  $\sim 3.7 \text{ \AA}$  of CNF-PEDOT:PSS solution cast on a TEM copper grid. The large crystalline structures are predominantly oval in shape and their size ranges from  $\sim 5 \text{ nm}$  up to  $\sim 25 \text{ nm}$  in width. On the other hand, neither the control pure PEDOT:PSS samples nor the CMC-PEDOT:PSS samples show any similar crystalline structures (Supplementary Figure 6). The size of most of the crystalline structures in Figure 5b is comparable to the size range observed in the AFM images of the PEDOT structures on top of the fibrils. However, it is not possible to identify the fibril in these images. Therefore, we suggest that these are PEDOT crystallites which were formed in the following steps: 1) PEDOT aggregates on the CNF surface in the wet state 2) the PEDOT aggregate detaches from the fibril with its structure intact in the wet state and finally 3) forms highly organized crystallites when dried.



**Figure 5.** TEM image of a) PEDOT:PSS on the surface of CNF and b) highly crystalline PEDOT:PSS segregated from CNF surface. Insets show magnified images of the selected areas.

Understanding the interactions between the constituents of a composite system and their impact on performance parameters are key steps towards building efficient devices. With the combined knowledge obtained from the different techniques it is possible to construct a reasonable model of the PEDOT organization in the presence of CNF and CMC.

For the CNF-PEDOT:PSS samples, the AFM scans show that the fibrils predominantly lie in the plane of the substrate, as expected. In addition, the AFM scans show that the PEDOT organizes itself in 13 nm bead-like structures, assembling itself around the fibrils, pointing to a strong ordering drive in the system. The GIWAXS transmission incidence data shows a symmetric  $\pi$ -stacking indicating that there are stacks of PEDOT in edge-on orientation with respect to the substrate surface. The parallel incidence GIWAXS shows a strong  $\pi$ -stacking peak at the  $0^\circ$  azimuthal angle indicating a face-on PEDOT stacking with respect to the substrate plane as well. The strong  $\pi$ -stacking and the weak lamellar organization indicate a strong PEDOT $\leftrightarrow$ CNF interaction in these samples as compared to the PEDOT $\leftrightarrow$ PEDOT interaction. The observations made are consistent with  $\pi$ -stacked PEDOT that covers the fibrils coaxially with a face-on orientation with respect to the surface of the fibrils. The TEM images which show stacks of crystals on the surface of the fibrils in the CNF-PEDOT:PSS samples are also in agreement with this model. The total surface area provided by the nano fibrils for PEDOT organization is several orders of magnitude larger than the surface area of the petri-dish substrate used to dry the self-standing films. Therefore, we conclude that the role of the petri-dish substrate in the PEDOT crystal formation is negligible.



**Figure 6.** Schematic model describing arrangement of PEDOT:PSS a) on CNF and b) with CMC

For the CMC-PEDOT:PSS samples, the surface available for the organization of the PEDOT crystals is the glass substrate onto which the film was cast. Hence, we expect the WAXS information to demonstrate the interaction of the PEDOT with the bottom of the free-standing film. The parallel incidence with the pronounced PEDOT lamellar peaks at the  $90^\circ$  azimuthal angle (Fig. 4d) indicates a strong edge-on  $\pi$ -stacking with respect to the substrate level. On the other hand, the 010 peak at  $90^\circ$  azimuthal angle indicates that there is also a considerable presence of  $\pi$ -stacked PEDOT in the face-on configuration. It is not clear why the PEDOT is found almost exclusively with edge-on and face-on orientations but not at other angles of orientation. Therefore, we speculate that the role of the substrate in the organization of the PEDOT extends throughout the bulk system.

PEDOT:PSS is seen to be highly malleable, forming conformal self-assembled structures on nanometer sized fibrils (Fig 6a). PEDOT:PSS, which has been described as a micelle,<sup>37</sup> is in this study observed to form spherical vesicles on, or around, suitably sized nano-structures in the

solution state. The nano-beads self-assemble on top of the fibrils in the wet state, as evident from the AFM sample preparation in which the PEDOT:PSS solution was drop cast on top of a pre-deposited CNF. This wet state aggregation of PEDOT is also in agreement with theoretical studies.<sup>59</sup> As previously suggested, the wet state aggregation can be followed by the formation of highly defined  $\pi$ -stacked crystals of PEDOT during the evaporation of the solvents.<sup>37, 43</sup> In the systems with the polymeric CMC, PEDOT:PSS is phase segregated into micelle-like PEDOT-rich structures embedded in a matrix of CMC and PSS.

Since the CNF-based and the CMC-based systems are markedly different from one another it is not trivial to construct a general theoretical model to describe their conductivity. However, by invoking available models for nano-particle conductors and two- and three-dimensional stick systems, we can qualitatively describe the observed difference in conductivity. The conductivity  $\sigma$  of interconnected conducting nanoparticles is given by the equation  $\sigma = \sigma_0(V_f - V_c)^s$  where  $\sigma_0$  is the conductivity of a single nanoparticle,  $V_f$  is the volumetric fraction of the nanoparticles,  $V_c$  is the percolation volumetric fraction and  $s$  is the critical exponent. The higher conductivity of the CNF-PEDOT:PSS correlates with a higher intrinsic conductivity of PEDOT nanoparticles due to improved  $\pi$ -stacking and lower percolation threshold due to high interconnectivity between nanoparticles on a single fibril. In addition, the theory of percolation thresholds in two- and three-dimensional stick systems,<sup>60, 61</sup> states that the threshold percolation concentration  $N_c$  of the nano-sticks is proportional to  $\frac{1}{L^2 r}$  when  $L \gg r$ , where  $L$  and  $r$  are the length and radius of the nano-stick, respectively. If we assume that the  $L$  in the CNF based system is the length of the fibril while the  $L$  in the CMC system is the length of a single, phase segregated PEDOT-rich region (see Figure 3f), we can expect that the CNF system with the high-aspect ratio fibrils will lead to a lower percolation threshold as compared to the CMC system. We also note that single-fibril conductivity

measurements would shed better light on whether it is possible to increase the conductivity of the CNF-PEDOT:PSS *power paper* in specific directions by an organized long-range alignment of the PEDOT:PSS coated CNFs in the paper.

Even though our study focused on cellulose:PEDOT:PSS systems, we note that similar nano-scale studies are indispensable for the full utilization of other bio-composite systems. This study shows the interaction between PEDOT:PSS and polymers or fibers. Hence, it is expected that other polymeric materials with similar structures will have similar properties to what is shown here,<sup>4</sup> and it is possible that self-organizing effects are present in other previously investigated fibrillated systems.<sup>5</sup> Therefore, we believe that the results presented here are extendable to other related systems. Furthermore, the observation that CNF promotes a better organization of PEDOT:PSS as compared to CMC in the presence of a secondary dopant, here DMSO, shows that CNF too can be understood to act as an additional secondary dopant alongside DMSO. Therefore, our study extends the concept of secondary-doping from a solvent-based mechanism to other material systems, here nano-fibrils.

## CONCLUSIONS

CNF proves to be a more versatile component in composite conducting papers, as compared to the common polymeric binder CMC. We have shown that, in addition to improving the mechanical stability, the nano-fibrils of CNF lead to high conductivity by virtue of the stronger surface interactions which create highly organized long-range conducting paths for charge transport. We used the direct imaging techniques AFM and TEM with specially adapted sample preparation steps to obtain images of PEDOT:PSS in a composite system with unprecedented clarity and resolution. These images confer a significant advance to our understanding of the nature of PEDOT:PSS crystallization. Furthermore, the structural organization was characterized with WAXS and

consistent models of PEDOT assembly on cellulose have been suggested. We show that several properties of the cellulose-based additives, such as structural, dimensional and chemical compatibility, influence the resulting properties of the composite film. Despite the difficulties inherent in the characterization of blend systems, this work brings attention to the necessity of understanding the fundamental intra-component interactions in formulating state-of-the-art bio-composites for a sustainable future.

## METHODS

**Materials.** CNF was prepared at RISE Bioeconomy from carboxymethylated softwood Kraft pulp by 10 passes through a high shear homogenizer at 700 MPa. Three different types of CNF with degrees of carboxymethyl substitution (DS) 0.06, 0.12, and 0.25 were prepared at 0.13 wt% in deionized water. The amount of charge was measured by conductometric titration. All CMCs (soluble non-fibrillated carboxymethylated cellulose) were obtained from Sigma Aldrich. The DS of the CMC was 0.76, 0.85, and 1.2. PEDOT:PSS with the product name PH 1000 (1.1 wt%) was purchased from Clevios. Polyethyleneimine, DMSO and glycerol were obtained from Sigma Aldrich. Mica substrate for AFM was obtained from Caspilor.

**Power paper.** Conducting papers were prepared as described previously by Malti et al.<sup>8</sup> In short the components cellulose: (PEDOT:PSS) : DMSO : glycerol were mixed in 1 : 2 : 11 : 1.6 dry weight ratio. For the CNF papers the solutions were mixed using a high shear mixer for 5 min while for the CMC the solutions were stirred with a magnetic stirrer. Each solution was poured on a petri dish and dried in an oven at 60 °C resulting in a flexible conducting paper.

**Electrical characterization.** For the conductivity measurements, the free-standing conducting paper was cut in small pieces of 4 X 20 mm and attached with conducting silver paste to a substrate



with evaporated gold 4-probes. A Keithley model 2400 was used to apply a current across the outer 2 probes while simultaneously measuring the resistance between the inner 2 probes from which the conductivity was calculated. The thickness of the conducting paper was measured by the surface profiler Dektak3ST manufactured by Veeco.

**Atomic force microscopy.** AFM investigations were performed with a Dimension 3100 model Veeco microscope utilizing Nanoscope software for data acquisition in tapping-mode configuration giving topography and phase images. The CNF samples were investigated on mica substrate. The mica was cleaned by the scotch-tape technique<sup>55</sup> and then a drop of 1g/L polyethylene imine (PEI) solution was put on it for 10 sec. The PEI solution was rinsed away with DI water and dried with N<sub>2</sub> gas, leaving a layer of PEI to anchor the CNF. This was followed by dropping 0.043 wt % of CNF solution on top of the PEI monolayer for 20 sec. Then the solution was rinsed with DI water, dried with N<sub>2</sub> gas and observed with AFM. For the PEDOT:PSS AFM samples, PH1000 was mixed with DMSO and glycerol at the ratio 1:0.02:0.005 (PH1000:DMSO:glycerol). A drop of this solution was also put for 50 seconds on top of the CNF lodged on mica and then rinsed off with DI water. The substrate was then heated on a hot plate at 100 °C for 5 min in order to remove excess solvents. CMC samples were prepared from a solution of CMC, PEDOT:PSS, DMSO and glycerol at the dry wt% ratio 1 : 2 : 11 : 1.6. This solution was spin casted at 1500 rpm onto native-Si substrates cleaned by UV-ozone treatment. The substrates were then baked at 100 °C for 5 min.

**Wide Angle X-ray Scattering.** In order to make the self-standing films used in the WAXS experiments, PEDOT:PSS, DMSO, and glycerol were mixed in the ratio (1:0.8:5.5) and homogenized using a shear mixer for 5 min. Following that the solution was dried in the oven at 70 °C giving free-standing films. WAXS measurements were done utilizing a Cu K-alpha X-ray

source emitting at 1.5418 Å, monochromatized and focused by 1D multilayer optics. The X-rays scattered from the sample were collected in vacuum using a Fuji photo-stimulable imaging plate in transmission and grazing incidence. Further details of the WAXS instrumentation can be found in Apitz et al.<sup>56</sup>

**TEM.** Investigations were done on standard lacey carbon film suspended by copper TEM grids (obtained from SPI Supplies). HRTEM imaging was performed using the Linköping double Cs corrected FEI Titan3 60-300, operated at 300 kV. Pure CNF samples with DS 0.12 at 0.04 wt % in water were dropped on the copper grid and rinsed with drops of water after 10 seconds. The copper grid was then placed on a hot plate at 100 °C for 3 min. CNF, PEDOT:PSS, DMSO were mixed at the dry wt% ratio 1 : 8 : 44 at 0.04 wt % in water and dropped on the copper grid, followed by rinsing with water after 10 seconds. The copper grid was then placed on the hot plate at 100 °C for 3 min.

### **Competing interests**

The authors declare no competing financial interests.

**Supporting Information:** The following files are available free of charge.

Additional AFM/TEM/WAXS data and a theoretical model of PEDOT:PSS organization on CNF. (PEDOTPSSonCellulose\_ACSAEM\_SuppInfo\_BelainehDetal.pdf)

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## Supporting Information

### Controlling the Organization of PEDOT:PSS on Cellulose Structures

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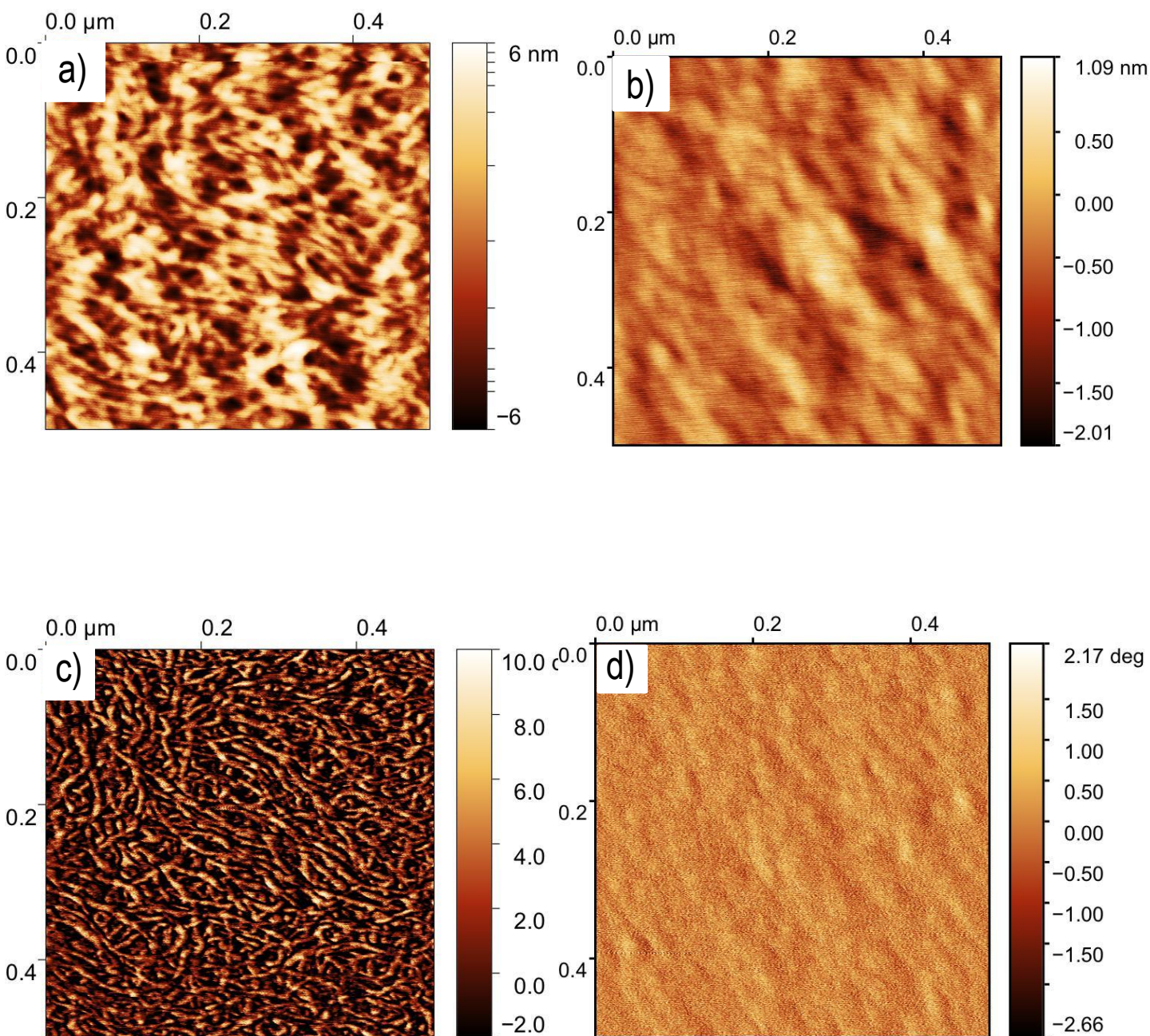
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5. RISE Bioeconomy

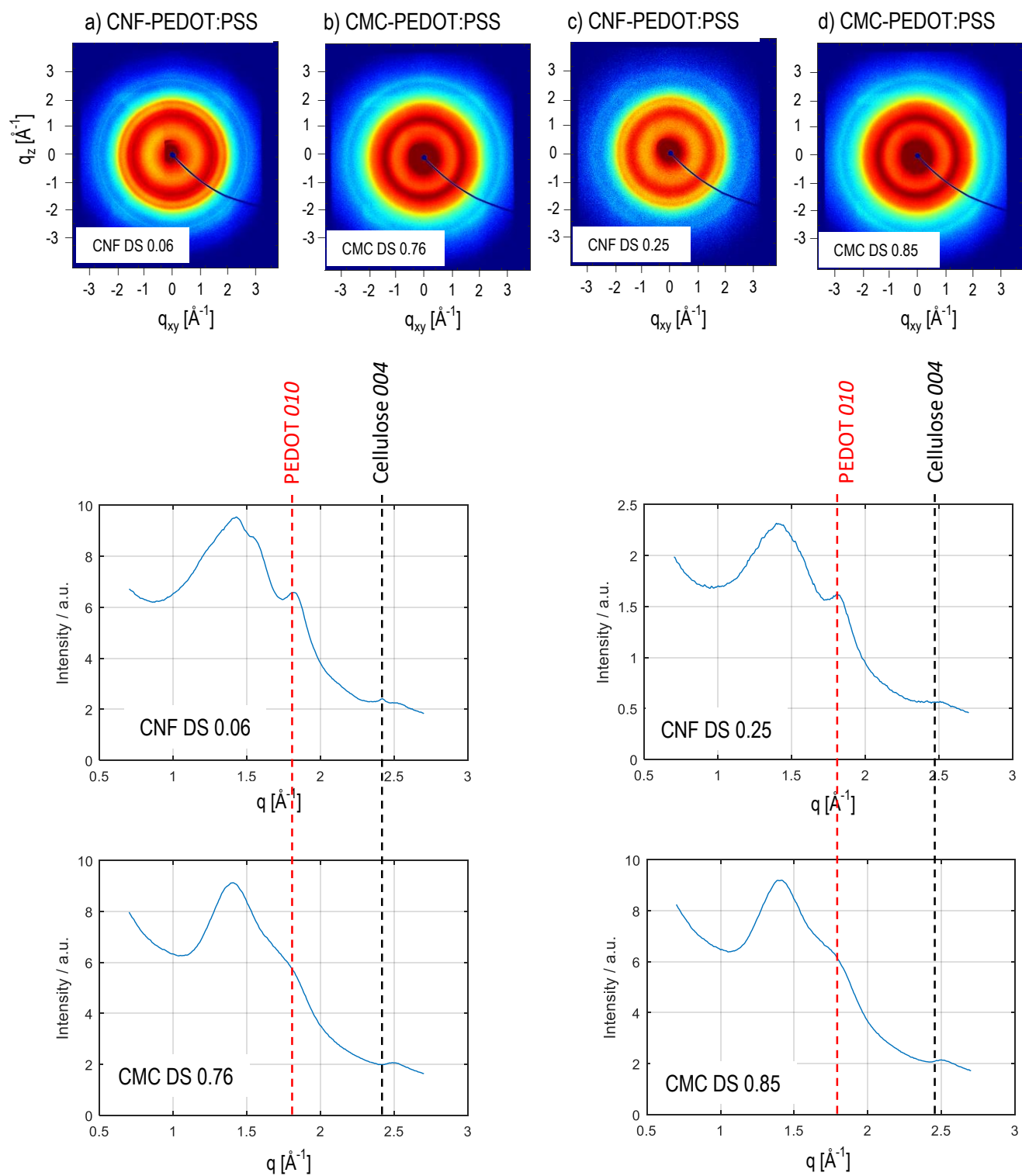
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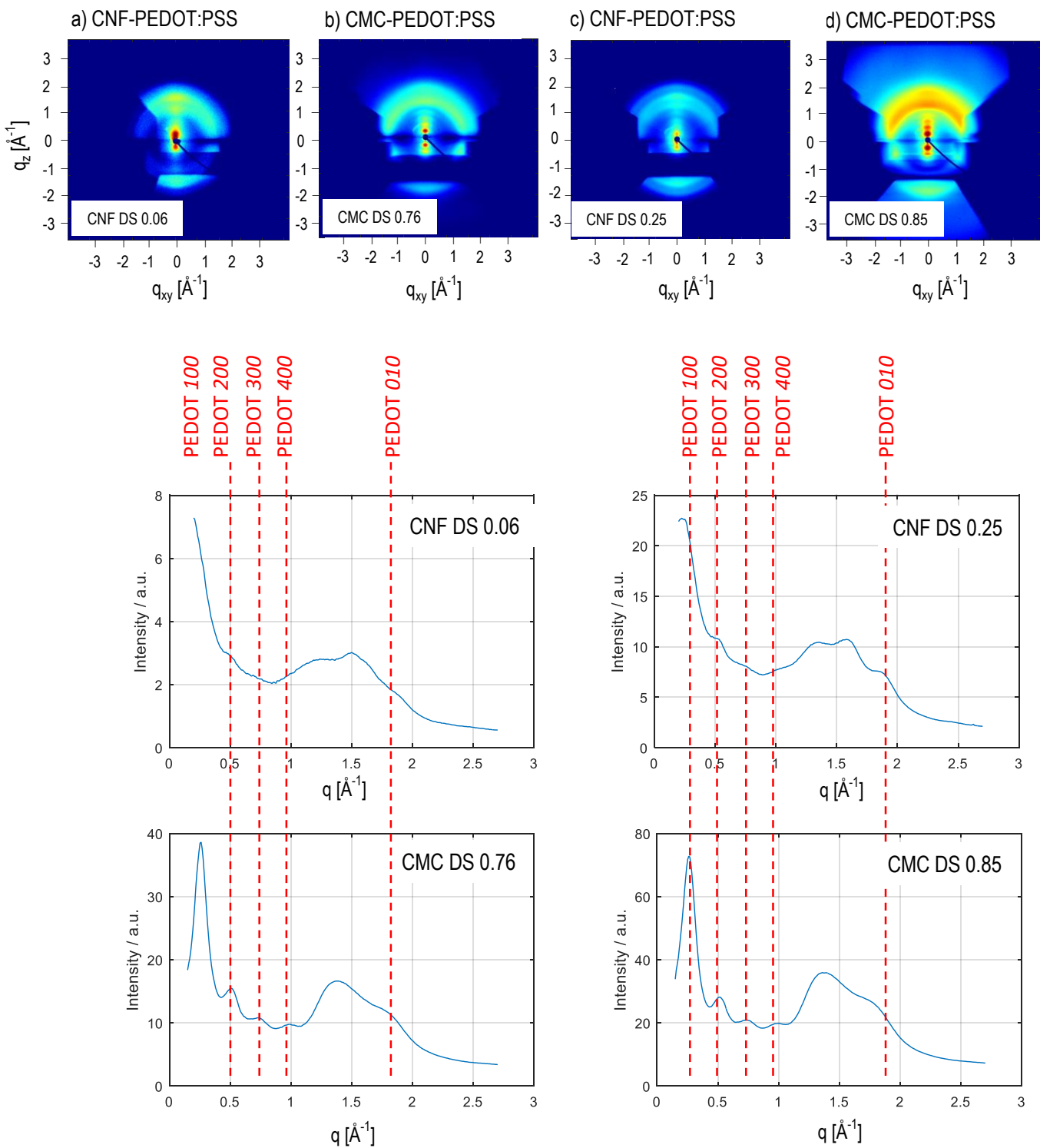
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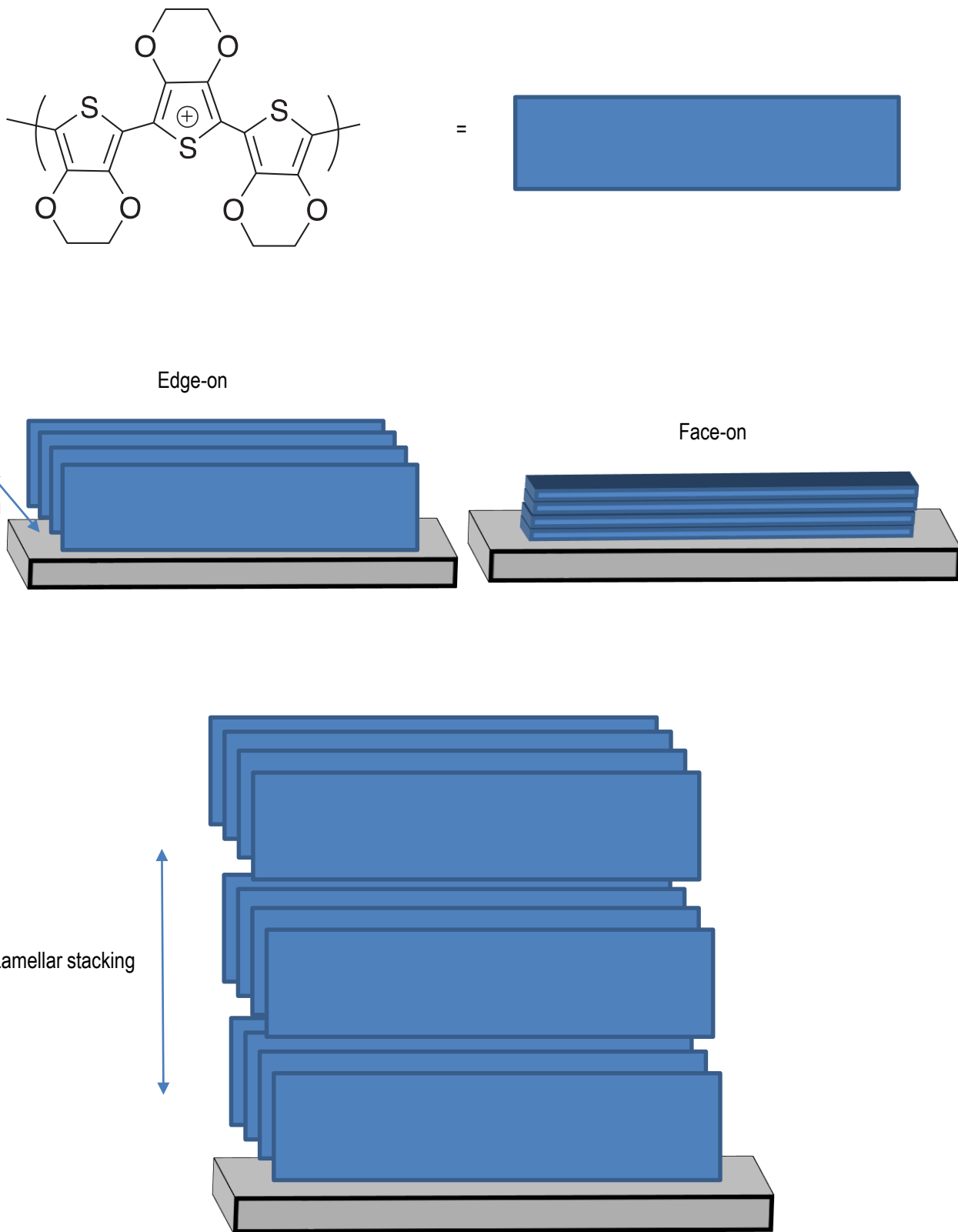
**Figure S1.** AFM topography and phase images of a) and c) PEDOT:PSS and b) and d) PSS-CMC, respectively.



**Figure S2.** WAXS in transmission mode for cellulose:PEDOT:PSS with different DS and the corresponding integrations.

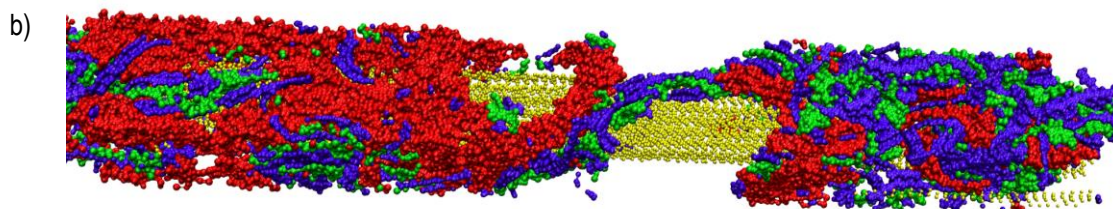
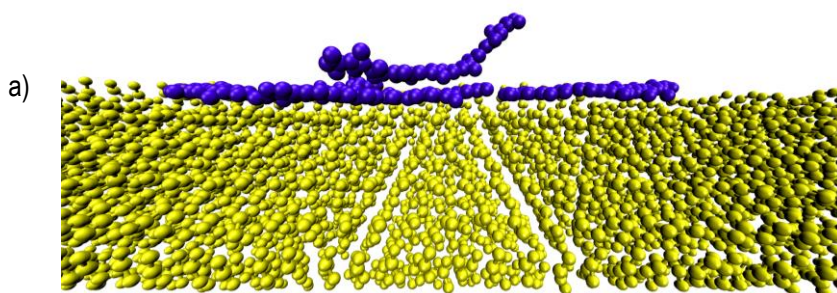


**Figure S3.** GIWAXS in parallel mode for cellulose:PEDOT:PSS with different DS and the corresponding integrations.

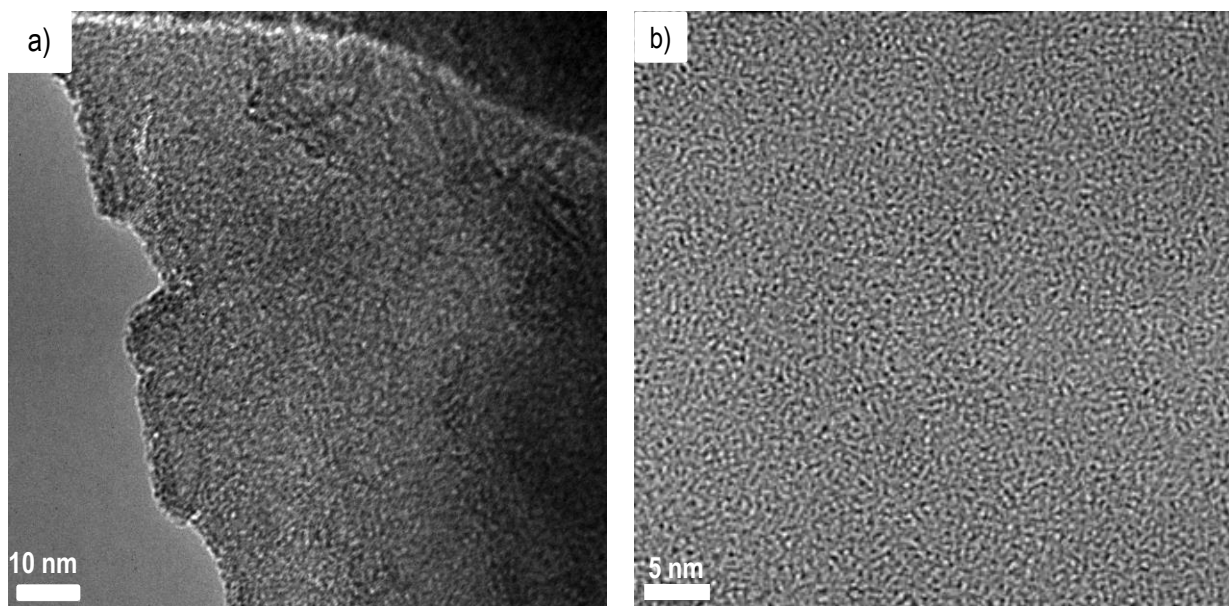


**Figure S4.** Schematic of the different modes of PEDOT stacking.





**Figure S5.** Molecular dynamics simulations of a) PEDOT (violet) face-on at the cellulose surface (yellow) and b) the whole simulation box including PEDOT (violet), PSS- (green), PSSH (red) and cellulose in yellow showing a predominantly face-on arrangement on the cellulose. The water is removed for clarity.



**Figure S6.** HRTEM of a) PEDOT:PSS and b) CMC-PEDOT:PSS.