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Structural and electronic properties of Li-intercalated graphene on SiC(0001)Nuala M. Caffrey,^{1,*} Leif I. Johansson,¹ Chao Xia,¹ Rickard Armiento,¹ Igor A. Abrikosov,^{1,2,3} and Chariya Jacobi¹¹*Department of Physics, Chemistry and Biology (IFM), Linköping University, SE-581 83, Linköping, Sweden*²*Materials Modeling and Development Laboratory, NUST "MISIS," 119049 Moscow, Russia*³*LACOMAS Laboratory, Tomsk State University, 634050, Tomsk, Russia*

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We investigate the structural and electronic properties of Li-intercalated monolayer graphene on SiC(0001) using combined angle-resolved photoemission spectroscopy and first-principles density functional theory. Li intercalates at room temperature both at the interface between the buffer layer and SiC and between the two carbon layers. The graphene is strongly n -doped due to charge transfer from the Li atoms and two π bands are visible at the \bar{K} point. After heating the sample to 300 °C, these π bands become sharp and have a distinctly different dispersion to that of Bernal-stacked bilayer graphene. We suggest that the Li atoms intercalate between the two carbon layers with an ordered structure, similar to that of bulk LiC₆. An AA stacking of these two layers becomes energetically favourable. The π bands around the \bar{K} point closely resemble the calculated band structure of a C₆LiC₆ system, where the intercalated Li atoms impose a superpotential on the graphene electronic structure that opens gaps at the Dirac points of the two π cones.

DOI: [10.1103/PhysRevB.93.195421](https://doi.org/10.1103/PhysRevB.93.195421)**I. INTRODUCTION**

The epitaxial growth of graphene on SiC substrates is a viable and attractive way of producing large homogeneous monolayer samples suitable for electronic device applications [1,2]. When grown on the Si terminated SiC(0001) surface, the first carbon buffer layer has a detrimental effect on the graphene charge carrier mobility and needs to be eliminated [3]. Intercalation of elements such as hydrogen [4,5], gold [6], germanium [7], silicon [8,9], nitrogen [10], fluorine [11], oxygen [12], lithium [13–15], and sodium [16–18] have all been shown to decouple the buffer layer and transform it into a quasi-free-standing graphene layer with varying degrees of doping depending on the intercalant. The alkali metals induce a strong n -type doping when deposited on graphene samples. Potassium [19,20], rubidium, and cesium [21] remain on the surface after deposition and do not penetrate to the interface even after heating. Lithium [13–15] and sodium [16–18], on the other hand, intercalate readily at room temperature, decoupling the buffer layer. Increasing the temperature to 300 °C (100 °C) promotes the complete intercalation of Li (Na), while higher temperatures result in the deintercalation and desorption of the metal from the sample. Interest in Li-intercalated graphene was originally motivated by the possibility of using it to improve the capacity of Li-ion batteries. However, the structural degradation of the carbon layers after Li deposition renders this unlikely [14]. More recently, superconductivity was discovered to occur in Li decorated monolayer graphene on SiC(0001) at low temperatures [22,23]. Detailed studies of the structural and electronic properties of Li intercalated graphene will be required in order to determine the origin and mechanism of this behavior.

Low-energy electron microscopy (LEEM) imaging provide evidence that Li atoms form small islands on the graphene surface directly after deposition [14]. Over time, these islands

coalesce, shrink and eventually disappear; this process can be accelerated by annealing. Core level spectra, combined with angle-resolved photoemission spectroscopy (ARPES) data, show that Li intercalates both at the interface between the substrate and the buffer layer and between the carbon layers, with the result that multiple π bands become visible [13,15]. A $(\sqrt{3} \times \sqrt{3})R30^\circ$ diffraction pattern was observed after Li deposition on monolayer graphene using low-energy electron diffraction measurements. This pattern did not appear when Li was deposited on a sample containing only the $(6\sqrt{3} \times 6\sqrt{3})R30^\circ$ carbon buffer layer, suggesting that it is due to an ordered intercalated Li layer between the two graphene sheets. The decoupling of the buffer layer was also shown to occur when Li was deposited on samples with an initial coverage of only the buffer layer. In this case, a single π band becomes visible directly after Li deposition [14], which can only occur if the buffer layer has been decoupled.

Several recent investigations have attempted to elucidate the exact nature of the Li intercalation on both zero- and monolayer graphene and its temperature dependence. Bisti *et al.* determined that the Li atoms occupy the T4 site of the topmost SiC bilayer after they intercalate underneath the buffer layer, thereby decoupling it from the substrate [24]. Sugawara *et al.* deposited Li on bilayer graphene at 30 K and reported the appearance of a sharp $(\sqrt{3} \times \sqrt{3})R30^\circ$ diffraction pattern [25], but did not observe this pattern after the deposition of Li on monolayer graphene. They suggested Li intercalates between the top two adjacent carbon layers and takes the same well-ordered superstructure as in bulk C₆Li. However, the band structure around the \bar{K} point for both Li-deposited monolayer and bilayer graphene contained only a single π band (see Figs. 2(e) and 2(f) in Ref. [25]). This contrasts significantly with earlier findings [13–15] where at least two π bands appear around the \bar{K} point after Li deposition on monolayer graphene.

We present here detailed ARPES data collected before and after Li deposition on monolayer graphene, both at room temperature and after heating to 300 °C, in order to unambiguously determine the number of π -band branches

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present and their dispersions. Directly after deposition, we show clearly the presence of multiple π bands around the \bar{K} point. After heating the sample to 300 °C, these bands become considerably sharper and their dispersions do not resemble that of Bernal stacked bilayer graphene [26,27]. To understand these observations, tight-binding and density functional theory band structure calculations were performed for Li intercalated free-standing bilayer graphene, as well as for Li-intercalated graphene systems that explicitly take the SiC(0001) surface into account.

Our combined experimental and theoretical results lead us to suggest the following: Directly after deposition, Li intercalates both underneath the buffer layer and between the two carbon layers. Heating the sample promotes the complete intercalation of the Li atoms to the interface, as well as the development of an ordered Li configuration between the two carbon layers. In the process, it becomes energetically more favorable for the carbon layers to become AA rather than Bernal-stacked. We show that the π bands visible around the \bar{K} point closely resemble the band structure of a C_6LiC_6 system as calculated with density functional theory. The periodic perturbation of the graphene electronic structure by the ordered Li layer induces “gaps” to open at the Dirac point of each of the π cones. Tight-binding calculations show that the interlayer coupling is enhanced by the presence of the Li atoms beyond what is typical in clean bilayer graphene, and the impact of the enhanced coupling terms on the band dispersion is verified by ARPES spectra.

II. METHODOLOGY

A. Experimental details

Monolayer graphene samples were grown *in situ* on *n*-type wafers of 4H-SiC(0001) purchased from SiCrystal, which were specified to have a misorientation error within 0.05°. ARPES measurements were performed using Beamline I4 on the MAX III ring at the MAX IV Laboratory, which is equipped with a spherical grating monochromator and a PHOIBOS 100 mm CCD analyzer from SPECS. The wide angular dispersion mode was selected, providing an acceptant angle of $\pm 14^\circ$. Each ARPES spectrum was collected parallel to the A - \bar{K} - A' direction of the Brillouin zone of graphene [see Fig. 1(j)], in steps of 0.25° along the $\bar{\Gamma}$ - \bar{K} - \bar{M} direction, using photon energies of 33 and 70 eV. From this data, the π -band structure along certain directions in the Brillouin zone can be determined, as well as the angular distribution pattern, $E_i(k_x, k_y)$, at selected initial state energies. A Ta foil was used as a reference to determine the Fermi level.

Deposition of Li was performed during a five minute interval using a commercial alkali metal source (from SAES Getters) and with the sample at room temperature. Further details can be found in Refs. [13–15]. Subsequent heating was carried out for four minutes at each selected temperature.¹ The sample temperature was determined using optical pyrometers.

¹Two sets of data were collected from two different monolayer graphene samples. For the first sample, the heating was carried out in steps at 8 different temperatures from 150 °C to 450 °C, while for the second sample the temperature was raised directly from room

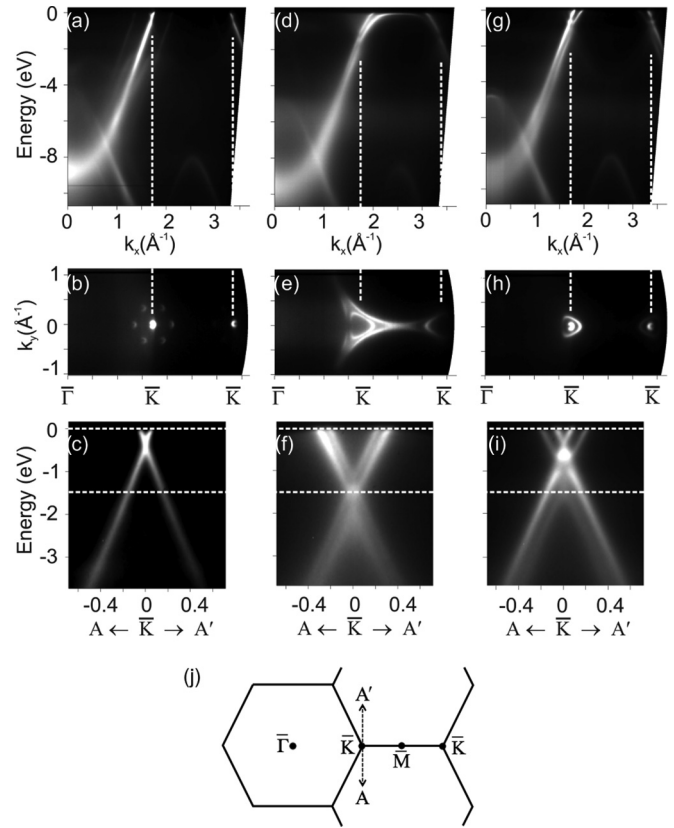


FIG. 1. Band dispersion recorded from monolayer graphene samples before [(a)–(c)] and after [(d)–(f)] Li deposition, and after heating to 300 °C [(g)–(i)]. The dashed lines are guides to the eye for illustrating the location of the first and second \bar{K} points or an initial state energy of 0 eV and -1.4 eV. The Fermi energy is located at 0 eV. (j) The 2D Brillouin zone of graphene showing the $\bar{\Gamma}$, \bar{M} , and \bar{K} high-symmetry points. The A and A' directions are also shown.

ARPES measurements were taken after the sample had cooled to room temperature. The Si 2*p* and Li 1*s* core level spectra were recorded before and after deposition, as well as after heating, and their peak areas were extracted. After Li deposition a Li 1*s*/Si 2*p* peak area ratio of around 1.6 was obtained which decreased to around 0.5 after heating to 300 °C. Using a simple layer attenuation model,² these ratios suggest the presence of three layers of Li in the surface region after Li deposition; one layer at the interface, one in

temperature to 300 °C. The band structure data obtained for both samples, both after Li deposition and after heating at 300 °C, were essentially identical.

²The following values were assumed when estimating the amount of Li in the surface region: a layer thickness of 3.45 Å for graphene and of 1.7 Å for Li, photoionization cross sections of 4.8 and 0.7 for Si 2*p* and Li 1*s* at 140 eV, respectively, concentration values of 0.5 for Si in SiC and 1 for Li in the Li layer, and an electron mean free path of 4.5 Å. The layer attenuation model then predicts a peak ratio of 1.88 when having a layer of Li at the interface, one in between the carbon layers and one on top of the graphene. When removing the Li layer on top of the graphene a ratio of 0.55 is predicted from the remaining layers at the interface and in between the carbon layers.

between the carbon layers and one on top of the graphene. After heating at 300 °C the ratio suggests that only the Li layers at the interface and between the carbon layers remain while the Li on top of graphene has disappeared, in agreement with earlier LEEM/PEEM results [14]. As the core level data showed the same features and changes upon heating as reported earlier [13], no core level data are included below.

B. Computational details

Density functional theory calculations were performed using VASP-5.3 [28–30]. The Perdew-Burke-Ernzerhof (PBE) [31] parametrization of the generalized gradient approximation (GGA) was employed. The plane-wave basis set was converged using an 800 eV energy cutoff. Structural relaxations of the C_6LiC_6 cell were carried out using a $9 \times 9 \times 1$ k -point Monkhorst-Pack mesh [32] to sample the Brillouin zone. A $24 \times 24 \times 1$ mesh was used to determine the total energies. The Tkatchenko-Scheffler method was used to describe van der Waals interactions [33]. The free-standing bilayer structures were optimized until all residual forces were less than $0.001 \text{ eV \AA}^{-1}$.

The SiC(0001) substrate was modelled using an asymmetric slab consisting of six bilayers of SiC(0001), arranged in the ABCACB stacking associated with the 6H polytype. A bulk termination was assumed. The GGA calculated lattice constant of bulk SiC is 3.09 \AA , in good agreement with the experimental value of 3.08 \AA . A vacuum layer of at least 15 \AA was included in the direction normal to the surface to ensure no spurious interactions between repeating slabs and the dipole correction was applied where appropriate. The dangling C bonds on the SiC(000 $\bar{1}$) surface were passivated with H atoms. The positions of the top two bilayers of SiC(0001), as well as the H-terminating atoms, the Li atoms, and all carbon layers, were optimized until all residual forces were less than 0.01 eV \AA^{-1} . The remaining atoms were held fixed at their bulk positions. The $(6\sqrt{3} \times 6\sqrt{3})R30^\circ$ surface reconstruction [34–37] of the buffer layer on SiC(0001) was modelled using the simplified $(\sqrt{3} \times \sqrt{3})R30^\circ$ cell which corresponds to a 2×2 graphene cell. Such an approximation was shown to be adequate to correctly describe the interaction between the SiC(0001) surface and the carbon layers [38].

III. RESULTS

A. Experimental ARPES spectra

The band structure obtained from the as-grown monolayer graphene sample, along the $\bar{\Gamma}-\bar{K}-\bar{M}$ direction of the Brillouin zone of graphene, is displayed in Fig. 1(a). The contribution from the π band dominates, with branches visible at both the first and second \bar{K} point. The photoelectron angular distribution pattern, $E_i(k_x, k_y)$, obtained at the Fermi energy is shown in Fig. 1(b). The image has been overexposed to show the presence of six replica π cones surrounding the Dirac cone at the first \bar{K} point, as well as the presence of the Dirac cone at the second \bar{K} point. The Fermi surface at the second \bar{K} point appears, when further exposing the image, as a segment of a circle [39,40]; it is a mirror image of the one at the first \bar{K} point. The lower intensity obtained around the second \bar{K} point is attributed to photoelectron

matrix effects [39,41]. The π -band structure obtained along the $A-\bar{K}-A'$ direction around the first \bar{K} point is shown in Fig. 1(c). A single, linearly dispersing π band is visible with the Dirac point located approximately 0.45 eV below the Fermi level. Taken together, the ARPES spectra in Figs. 1(a), 1(b), and 1(c) demonstrate that high quality, monolayer graphene is present on the SiC(0001) surface [26].

Depositing Li on the sample at room temperature induces significant changes in the band structure, as illustrated in Figs. 1(d), 1(e), and 1(f). Figure 1(d) shows that the entire π -band structure, here displayed in the $\bar{\Gamma}-\bar{K}-\bar{M}$ direction, is rigidly shifted downwards by about 1 eV; the position of the π band at the $\bar{\Gamma}$ point is now located at -10 eV , whereas for the clean sample, it is located at approximately -9 eV [cf. Fig. 1(a)]. A second π band is now visible close to the Fermi energy and the dispersion of these bands is no longer linear. The lower branch is instead quite flat and located below the Fermi level. This can be seen clearly in the Fermi surface [Fig. 1(e)] where the π bands are shown to be occupied all the way between the first and second \bar{K} points. Again, the Fermi surface near the second \bar{K} point is a mirror image of that near the first; this is visible when the image exposure is increased. The π bands in the $A-\bar{K}-A'$ direction are shown in Fig. 1(f). The Dirac point has now shifted down in energy to approximately 1.4 eV below the Fermi level. This is due to the strong n -type electron doping provided by the Li atoms to the graphene layers.

Heating the sample to 300 °C induces further changes in the band structure, as illustrated in Figs. 1(g), 1(h), and 1(i). The entire band structure, displayed in Fig. 1(g), is now rigidly shifted by about 0.8 eV back towards the Fermi level. This can be seen by comparing the π bands at the $\bar{\Gamma}$ point in (g) and (d). Two π -band branches are now clearly resolved in an energy window between the Fermi level and -6 eV . The Fermi surface in Fig. 1(h) shows that neither of these π -band branches is continuously located below the Fermi level between the first and second \bar{K} point. Again, the Fermi surfaces at the first and second \bar{K} points are mirror images of each other.

Comparing Figs. 1(i) and 1(f), one can see that the π bands around the \bar{K} point are considerably sharper, and their dispersions distinctly different, after heating than directly after Li deposition at room temperature. The Dirac point has moved to approximately 0.6 eV below the Fermi level with two distinct π bands visible above and below it. The bands show neither the parabolic dispersion of Bernal stacked bilayer graphene nor the linear dispersion predicted for AA stacking [26,27].

Photoemission angular distribution patterns extracted at several fixed energies above and below the Dirac point are shown in Fig. 2. At energies far from the Dirac point, namely at -0.1 and -1.5 eV from the Fermi energy, two ringlike patterns corresponding to the two π bands are present. A strong triangular deformation, or trigonal warping, of the outer Dirac cone is clearly visible at these energies. At energies closer to the Dirac point, for example, at -0.4 and -1.1 eV , the trigonal warping is considerably reduced and the inner ring has been reduced to a point. Finally, the angular distribution at energies of -0.6 and -0.8 eV are not comprised of ringlike segments but of two and three points of varying intensity, respectively. In all cases, the intensity anisotropy of the angular maps does

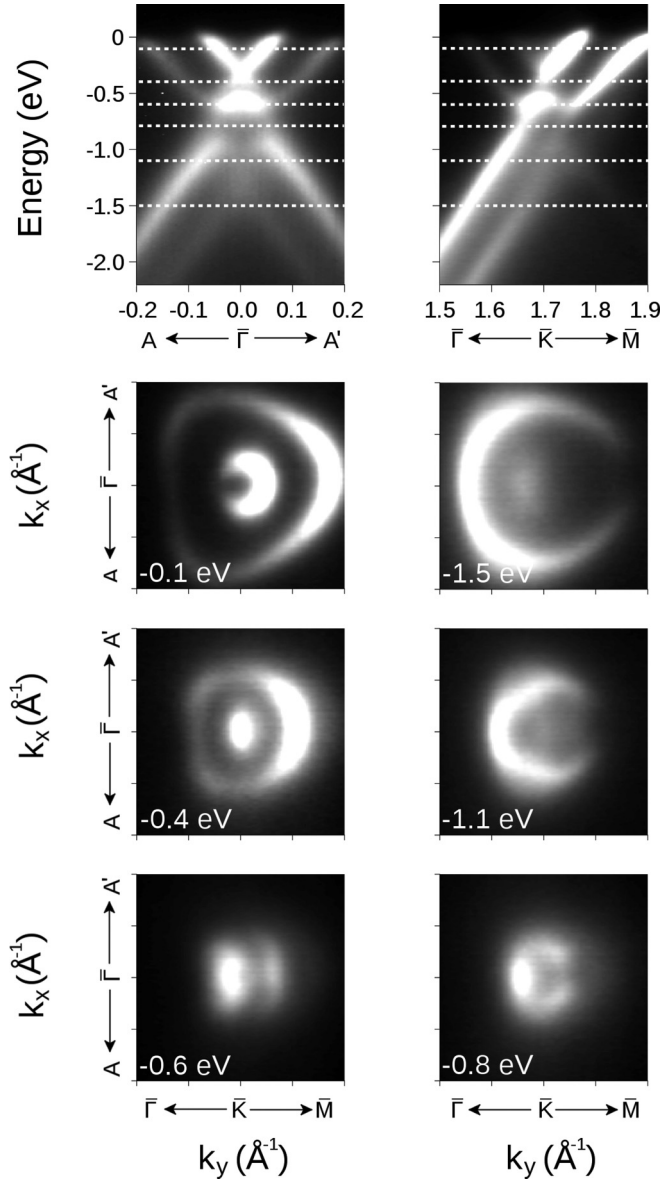


FIG. 2. Band dispersion recorded in the vicinity of the \bar{K} point (top two panels) and angular distributions extracted at fixed energies close to the Dirac point. The six energies are shown as white dashed lines through the band dispersions in the top panels. The Fermi energy is located at 0 eV. An incident photon energy of 33 eV was used.

not correspond to that predicted for Bernal-stacked bilayer graphene [39,40].

B. DFT band structures

The ARPES spectra presented in the previous section shows clearly the presence of two π bands after Li deposition, which implies that Li has intercalated to the interface and decoupled the buffer layer from the substrate. This has previously been corroborated by both experimental core-level spectra and DFT calculations and verified in the current work (see Appendix). We will now address the relationship between the presence of Li between the two carbon layers and its effect on the dispersion by calculating the band structure of free-standing

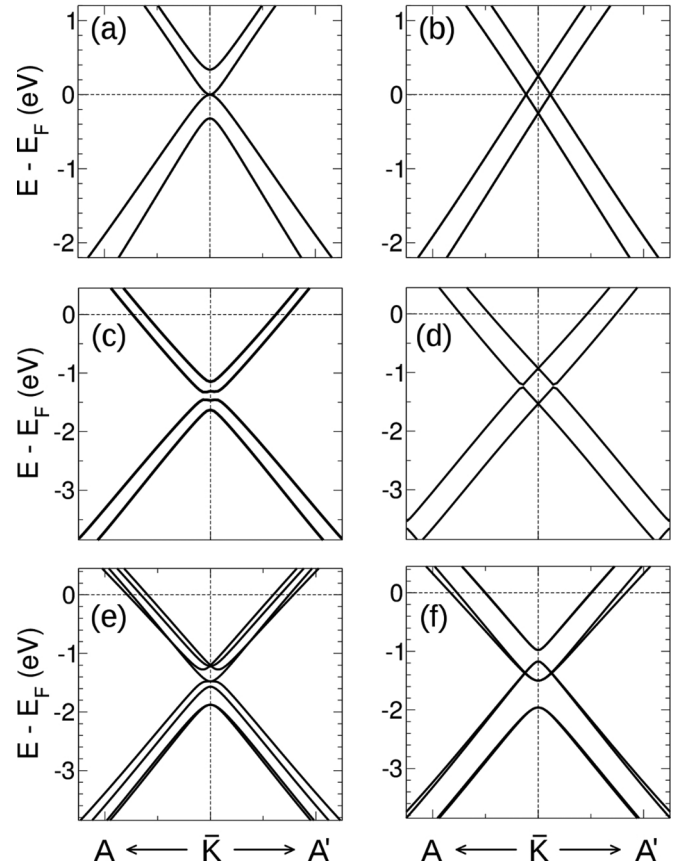


FIG. 3. Calculated band structure of (a) AB-stacked clean bilayer graphene, (b) AA-stacked clean bilayer graphene (c) AB-stacked C_8LiC_8 , (d) AA-stacked C_8LiC_8 , (e) AB-stacked C_6LiC_6 , and (f) AA-stacked C_6LiC_6 . The high-symmetry labels are those of the graphene unit cell.

bilayer graphene with and without intercalated Li, and with different graphene stackings.

We first consider the case where Li does not intercalate between the two carbon layers. These two layers then remain Bernal stacked and the resulting band structure has a symmetric, parabolic band dispersion around the \bar{K} point, as shown in Fig. 3(a). The band structure of clean AA-stacked bilayer graphene, shown in Fig. 3(b), is comprised of two Dirac cones with linear dispersion, separated in energy by 0.5 eV.

We then consider the case where Li intercalates between the two carbon layers, which remain AB-stacked, such that a Li atom sits on top of the center of a hexagon for one graphene layer and directly underneath a carbon atom of the second carbon layer. The band structure associated with such a scenario is shown in Fig. 3(c) for a C_8LiC_8 concentration. The structural asymmetry is evident in the resulting “Mexican hat” band dispersion. A gap opens at the \bar{K} point as a result of the asymmetric doping of the two carbon layers. For this particular Li concentration a gap of 0.13 eV is opened. The third possibility is that the registry between the two graphene layers switches from AB to AA stacking. A sufficiently high concentration of intercalated Li atoms is already known to shift the stacking pattern of bilayer graphene and Li-graphite intercalation compounds (Li-GICs) to an AA-stacking of the

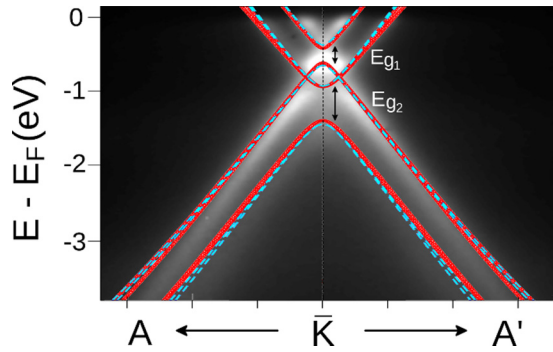


FIG. 4. Band structure of free-standing C_6LiC_6 as calculated with DFT (red solid lines) and using a tight-binding model (blue dashed lines), compared to the experimental ARPES spectra [cf. Fig. 1(i)]. The calculated band structures have been shifted rigidly in energy to match the top π^* band.

carbon sheets [42,43]. The resulting band structure is shown in Fig. 3(d). It has the linear dispersion typical of AA-stacked bilayer graphene. The effect of the intercalated Li can be seen in the increased energy separation in the two Dirac cones, from 0.5 to 0.6 eV.

The highest concentration of Li found in GICs at ambient conditions is LiC_6 . The $(\sqrt{3} \times \sqrt{3})R30^\circ$ diffraction pattern observed after Li deposition at room temperature [13] would suggest that this Li concentration is also the most favourable in bilayer graphene. If we consider AB-stacked graphene with a C_6LiC_6 concentration, we find three π bands are visible, due to the asymmetric doping of the two graphene layers, as shown in Fig. 3(e). Finally, we consider AA-stacked graphene at the same concentration. The energy dispersion, shown in Fig. 3(f), is comprised of two Dirac cones shifted relative to one another by the interlayer coupling and with a gap opened in both at the \bar{K} point. We find that AA-stacked C_6LiC_6 is 36 meV/C lower in energy than the equivalent AB-stacked configuration. As a comparison, Bernal stacked clean bilayer graphene is 6 meV/C lower in energy than AA-stacked bilayer graphene.

Comparing the band structures presented in Fig. 3 with the experimental ARPES spectra in Fig. 1(i), we find that good agreement is only achieved if we assume a C_6LiC_6 concentration of Li atoms between AA-stacked bilayer graphene. This is shown in Fig. 4. The development of severe cracks and wrinkles after Li deposition [14] further supports this postulation. We suggest that this is due to the strain induced in the system when the top carbon layer attempts to shift to an AA stacking. These cracks do not appear when Li is deposited on the buffer layer only.

C. Tight-binding model

We will now discuss the unusual shape of the electronic band dispersion close to the Dirac point, and in particular the opening of gaps at the \bar{K} point of the two Dirac cones, using a simple tight-binding model. The Li atoms, which intercalate in an ordered fashion, cause a periodic modification of the graphene electronic structure. The effect can be described

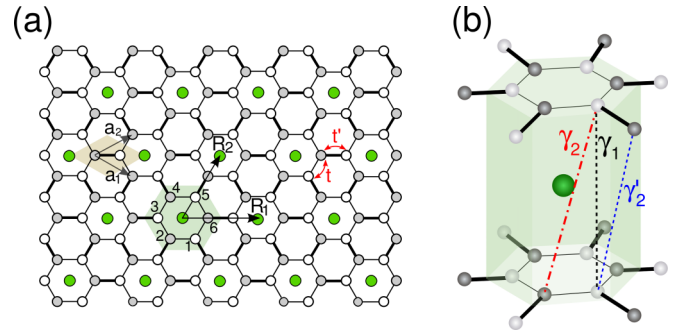


FIG. 5. Structure of C_6LiC_6 . (a) Top view showing the graphene lattice vectors \mathbf{a}_1 and \mathbf{a}_2 and the lattice vectors of C_6LiC_6 , \mathbf{R}_1 and \mathbf{R}_2 . \mathbf{R}_1 and \mathbf{R}_2 are rotated by 30° with respect to \mathbf{a}_1 and \mathbf{a}_2 and are $\sqrt{3}$ longer. The two carbon sublattices are denoted with gray and white circles and Li atoms are shown as green circles. The two types of bonds associated with the Kekulé textured graphene are shown as thin and thick solid lines. The two different intralayer hopping parameters between carbon atoms associated with these two bonds are denoted t and t' . The six atoms belonging to the C_6LiC_6 unit cell of one of the graphene layers are labeled 1–6, and are used to generate the Hamiltonian of the system. (b) Side view showing the interlayer coupling parameters, γ_1 between carbon atoms directly on top of one another, and γ_2 and γ_2' describing the textured skew coupling terms.

by a periodic potential $U(\mathbf{r}) = U(\mathbf{r} + l_1\mathbf{R}_1 + l_2\mathbf{R}_2)$, where $\mathbf{R}_1 = n_1\mathbf{a}_1 + m_1\mathbf{a}_2$, $\mathbf{R}_2 = n_2\mathbf{a}_1 + m_2\mathbf{a}_2$, and \mathbf{a}_1 and \mathbf{a}_2 are the graphene lattice vectors [44]. l , n , and m are integers. In our case, \mathbf{R}_1 and \mathbf{R}_2 describe the lattice vectors of C_6LiC_6 as shown in Fig. 5(a). Here, $\mathbf{a}_1 = (\frac{\sqrt{3}}{2}, -\frac{1}{2})a_0$, $\mathbf{a}_2 = (\frac{\sqrt{3}}{2}, \frac{1}{2})a_0$, $\mathbf{R}_1 = (\sqrt{3}, 0)a_0$, and $\mathbf{R}_2 = (\frac{\sqrt{3}}{2}, \frac{3}{2})a_0$, where a_0 is the graphene lattice constant. A gap will open at the \bar{K} point when the periodic potential in reciprocal space, $U(\mathbf{K})$, is not equal to zero. This condition will be satisfied when $(n_1 - m_1) \bmod 3 = 0$ and $(n_2 - m_2) \bmod 3 = 0$. This rule is reminiscent of that describing the gap chirality in carbon nanotubes [45] and the effect has already been harnessed to open band gaps in monolayer and bilayer graphene through the periodic patterning of antidot lattices [46,47]. The triangular symmetry associated with a $(\sqrt{3} \times \sqrt{3})R30^\circ$ Li adsorption pattern has already been shown to satisfy the conditions necessary to open a gap in monolayer graphene at the Dirac point [48,49].

We will now extend the model of Ref. [49] from Li decorated monolayer graphene to AA-stacked bilayer C_6LiC_6 graphene using a tight-binding model that takes the effect of the intercalated Li into account via a Kekulé texturing of the nearest-neighbor hopping parameters. The 12×12 matrix that describes the Hamiltonian of such a system has the following form:

$$H = \begin{pmatrix} H_T & H_{TB} \\ H_{BT} & H_B \end{pmatrix},$$

where $H_{T(B)}$ is the 6×6 matrix describing intralayer hopping in the top (bottom) graphene layer and H_{TB} is the 6×6 matrix describing interlayer coupling. The intralayer Hamiltonian,

$H_T = H_B$, is given by

$$\begin{pmatrix} \epsilon & t & 0 & t'e^{i\vec{k}\cdot\vec{\tau}_1} & 0 & t \\ t & \epsilon & t & 0 & t'e^{-i\vec{k}\cdot\vec{\tau}_2} & 0 \\ 0 & t & \epsilon & t & 0 & t'e^{i\vec{k}\cdot\vec{\tau}_3} \\ t'e^{-i\vec{k}\cdot\vec{\tau}_1} & 0 & t & \epsilon & t & 0 \\ 0 & t'e^{i\vec{k}\cdot\vec{\tau}_2} & 0 & t & \epsilon & t \\ t & 0 & t'e^{-i\vec{k}\cdot\vec{\tau}_3} & 0 & t & \epsilon \end{pmatrix},$$

where t and t' are the textured nearest-neighbor carbon hoppings [see Fig. 5(a)], ϵ is the on-site energy and $\tau_{1,2} = a_0(\frac{\sqrt{3}}{2}, \mp \frac{3}{2})$ and $\tau_3 = a_0(-\sqrt{3}, 0)$. The interlayer coupling matrix, $H_{TB} = H_{BT}$, is given by

$$\begin{pmatrix} \gamma_1 & \gamma_2 & 0 & \gamma'_2 e^{i\vec{k}\cdot\vec{\tau}_1} & 0 & \gamma_2 \\ \gamma_2 & \gamma_1 & \gamma_2 & 0 & \gamma'_2 e^{-i\vec{k}\cdot\vec{\tau}_2} & 0 \\ 0 & \gamma_2 & \gamma_1 & \gamma_2 & 0 & \gamma'_2 e^{i\vec{k}\cdot\vec{\tau}_3} \\ \gamma'_2 e^{-i\vec{k}\cdot\vec{\tau}_1} & 0 & \gamma_2 & \gamma_1 & \gamma_2 & 0 \\ 0 & \gamma'_2 e^{i\vec{k}\cdot\vec{\tau}_2} & 0 & \gamma_2 & \gamma_1 & \gamma_2 \\ \gamma_2 & 0 & \gamma'_2 e^{-i\vec{k}\cdot\vec{\tau}_3} & 0 & \gamma_2 & \gamma_1 \end{pmatrix},$$

where γ_1 is the vertical interlayer coupling between two carbon atoms directly on top of one another and γ_2 and γ'_2 are the textured skew hopping terms. These are sketched in Fig. 5(b).

The resulting band structure is shown by blue dashed lines in Fig. 4. We find that, as a result of the intercalation symmetry, gaps open at the \bar{K} point of each Dirac cone. These two cones are then shifted in energy relative to one another with an energy equal to the magnitude of the vertical interlayer coupling, $|\gamma_1| = 0.34$ eV. The in-plane hopping parameters, $t = 2.79$ eV and $t' = 2.59$ eV, agree with those found in Ref. [49] to describe Li-adsorbed monolayer graphene. The values of $\gamma_1 = -0.34$ eV and $\gamma'_2 = 0.04$ eV are characteristic of AA-stacked bilayer graphene and graphite [50].

In contrast to clean AA-stacked graphene or graphite, however, the effect of the skew coupling terms, γ_2 and γ'_2 , cannot be neglected. The effect of their textured nature on the band structure can be seen in the breaking of the equivalence of the gaps at the \bar{K} point of each Dirac cone. If γ_2 and γ'_2 were equal, the gap opened in each cone would also be equal. Instead we have that one gap, E_{g_1} , is 0.20 eV wide, while the other, E_{g_2} , is 2.3 times larger, at 0.46 eV. This ratio is approximately equal to that between γ_2 and γ'_2 , which are 0.10 and 0.04 eV, respectively.

To conclude, we have shown that Li intercalation serves to switch the stacking in a graphene bilayer from AB to AA. The highly-ordered Li layer generates a periodic perturbation of the graphene electronic structure such that a gap opens in each cone at the Dirac point. The increase in magnitude of the skew coupling terms is significant and its effects are visible in the experimental ARPES spectra.

IV. CONCLUSION

We have carried out detailed ARPES (angle-resolved photoelectron spectroscopy) studies of the band structure of monolayer graphene samples, before and after Li deposition at room temperature as well as after subsequent heating to 300 °C. Li intercalation is shown to have a large impact on the dispersion of the bands at energies close to the Dirac point, as well as donating charge to the graphene layers. Directly after

Li deposition we observe the appearance of a second π band. Upon heating to 300 °C the electronic band structure changes significantly: the π bands become sharper and have a distinctly different dispersion to that of Bernal stacked bilayer graphene. To understand these observations, we performed density functional theory band structure calculations for Li intercalation in free-standing bilayer graphene for different stacking of the layers. Our combined experimental and theoretical results show that, after Li deposition and subsequent heating, Li both intercalates underneath the buffer layer—decoupling it from the substrate—and between the two carbon layers. In the process, it becomes energetically more favorable for the carbon layers to become AA-stacked rather than Bernal stacked. We show that the π bands around the \bar{K} point closely resemble the calculated band structure of a C_6LiC_6 system, where the intercalated Li atoms introduce a periodic perturbation to the graphene electronic structure that opens gaps at the \bar{K} point of the two Dirac cones.

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APPENDIX A: LI-INTERCALATED GRAPHENE ON SiC(0001)

To determine the effect of doping due to the substrate, and also to verify that Li intercalation is sufficient to decouple the carbon buffer layer from the Si surface, we now take the SiC(0001) surface into account explicitly. Due to computational constraints, we could not model a C_6LiC_6 bilayer of graphene on SiC(0001) and therefore cannot reproduce the gaps that open at the \bar{K} point of the two Dirac cones as a result of this periodic potential. Instead, we model a C_8LiC_8 concentration. In agreement with the results of Refs. [24] and [51], we find that Li intercalation at the interface is sufficient to decouple the carbon buffer layer from the substrate. The Li atom is located at the T4 site on the SiC(0001) surface, breaking the Si–C bonds. The decoupled graphene sheet is then free to move laterally so that the center of the carbon hexagon is directly over the Li atom beneath it.

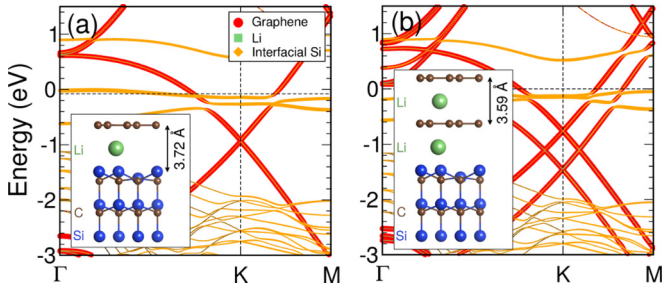


FIG. 6. Electronic band structure of (a) the Li intercalated zero layer graphene on SiC(0001) system and (b) Li intercalated monolayer graphene on SiC(0001). The Li concentration corresponds to LiC_8 and LiC_8LiC_8 , respectively. The insets show the relaxed structure of both configurations. The Fermi energy is located at 0 eV.

The band structure for the configuration that includes only zero layer graphene is shown in Fig. 6(a). A single π band, associated with the now decoupled carbon layer, is visible. The Dirac point is located 0.93 eV below the Fermi energy due to the strong doping from the Li atom. Figure 6(b) then shows the band structure of the configuration that includes a second carbon layer. Li atoms are now present both underneath the buffer layer and between the two carbon layers. As in the free-standing case, the AA-stacked configuration is energetically preferred. Two Dirac cones are now visible, separated in energy by 0.7 eV. This would suggest an interlayer coupling, γ_1 , of approximately 0.35 eV, similar to what we found for the free-standing intercalated bilayer. As a result of the doping due to these Li atoms, the Dirac points of the two cones are located 1.5 and 0.8 eV below the Fermi level, respectively.

APPENDIX B: REDUCTION IN DOPING AFTER HEATING

Finally, we discuss the observation that the graphene doping level decreases significantly, from -1.4 to -0.6 eV, after heating to 300°C [cf. Figs. 1(f) and 1(i)]. Core-level spectra shows that this dramatic reduction in doping occurs despite evidence that almost half of the Li atoms have not deintercalated or desorbed from the sample. The Li $1s$ core level spectrum shows that the component assigned to surface-adsorbed Li essentially disappears while the components assigned to intercalated Li remain, in agreement with earlier observations [13–15,24]. We therefore suggest that some Li atoms which were initially adsorbed on the surface, or between

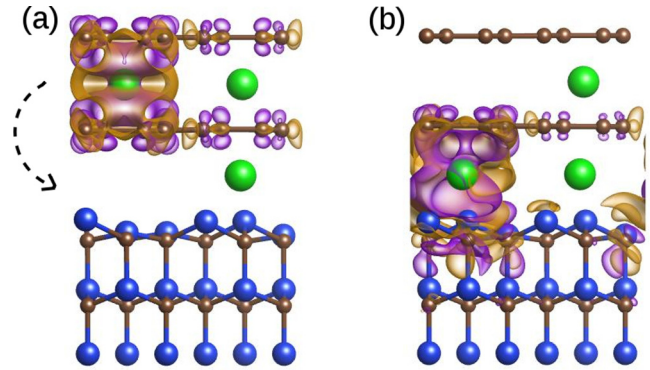


FIG. 7. Side view of a 2×1 unit cell of SiC(0001) that includes two layers of graphene (16 C atoms in each) and three Li atoms. Isosurfaces of charge density difference show how charge is transferred from a Li atom inserted (a) between the two carbon layers and (b) at the interface between SiC and the bottom carbon layer. A purple (orange) isosurface refers to a gain (loss) in charge density. The isosurface value is 0.0005 e/a.u.^3 .

the two graphene layers, intercalate to the interface between SiC and graphene after heating. In doing so, part of the charge which was previously transferred to the carbon atoms is now transferred to the SiC substrate, thus reducing the graphene doping level. If we consider a 2×1 SiC(0001) unit cell, that includes two graphene layers and three Li atoms, as sketched in Fig. 7, we find that it is energetically more favourable, by 0.54 eV, for one of the Li atoms to move from between the two graphene layers to the interface between the substrate and the bottom graphene layer.

To visualize the charge transfer, we show in Fig. 7(a) the charge density difference (CDD) that occurs when a Li atom is inserted between the top two carbon layers. The CDD is defined as $\Delta\rho = \rho_{\text{final}} - \rho_{\text{initial}} - \rho_{\text{Li}}$, where ρ_{final} , ρ_{initial} , and ρ_{Li} are the charge densities of the final system, the initial system that includes only two Li atoms, and an isolated Li atom, respectively. The charge is transferred equally between the two carbon layers. In contrast, Fig. 7(b) shows how the charge is distributed from the Li atom when it is placed at the interface between SiC and the buffer layer. In this case, the majority of the charge is transferred to the substrate, with only a small amount transferred to the decoupled buffer layer. This process can explain the shifting of the Dirac cones towards the Fermi level after heating as observed in the ARPES spectra.

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